

African School and Workshop

on X-Rays in Materials

some established techniques and practical applications

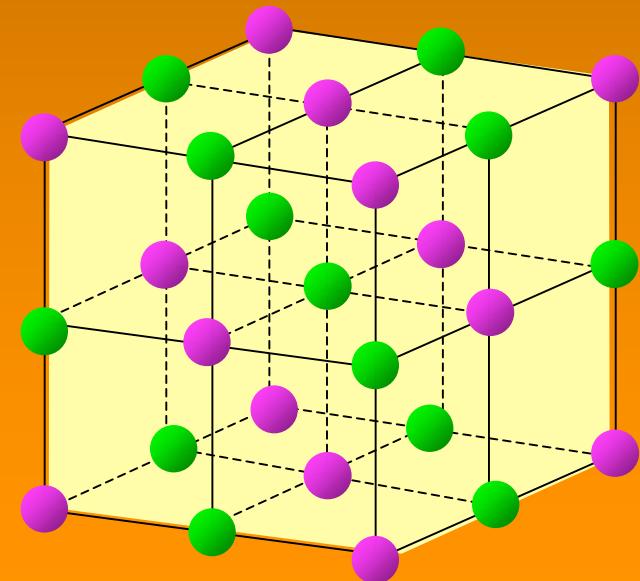
Dakar, Senegal
December 12-17, 2005



Basic crystallography



Paolo Fornasini
Department of Physics
University of Trento, Italy



African School and Workshop

on X-Rays in Materials

some established techniques and practical applications

Dakar, Senegal
December 12-17, 2005



Overview

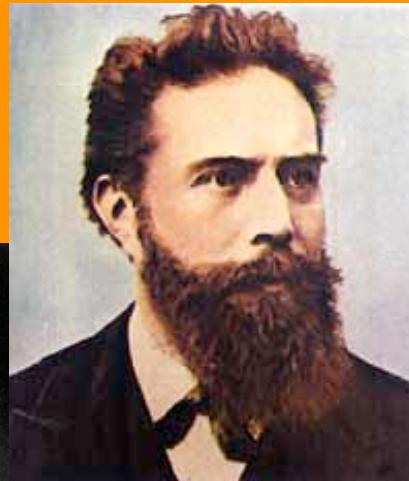
- X-rays
- Crystals
- Crystal lattices
- Some relevant crystal structures
- Crystal planes
- Reciprocal lattice
- Crystalline and non-crystalline materials



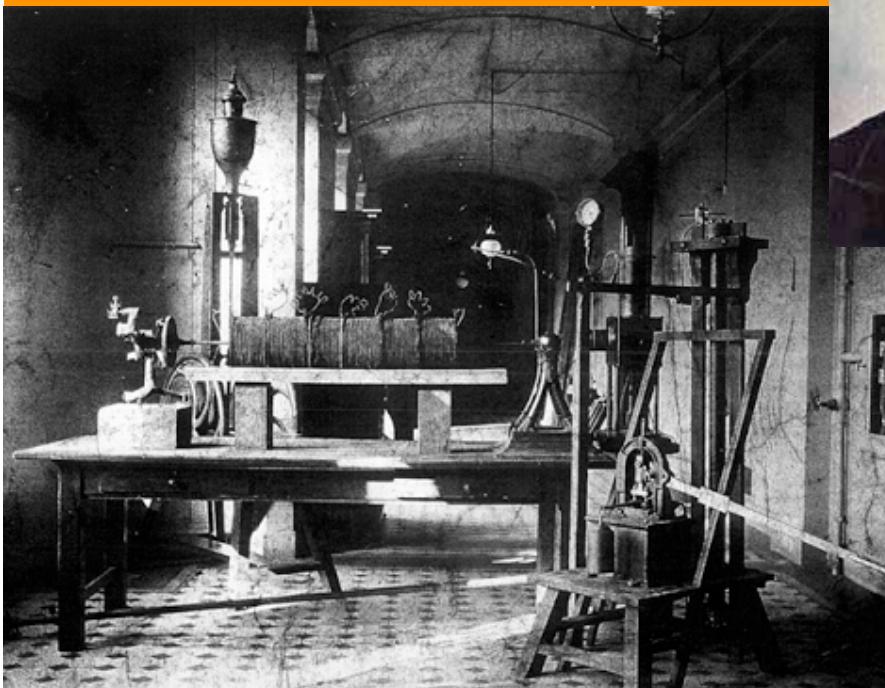
X-rays

1895 - Discovery of X-rays

Paolo
Fornasini
Univ. Trento



Wilhelm Konrad Röntgen
(1845-1923)



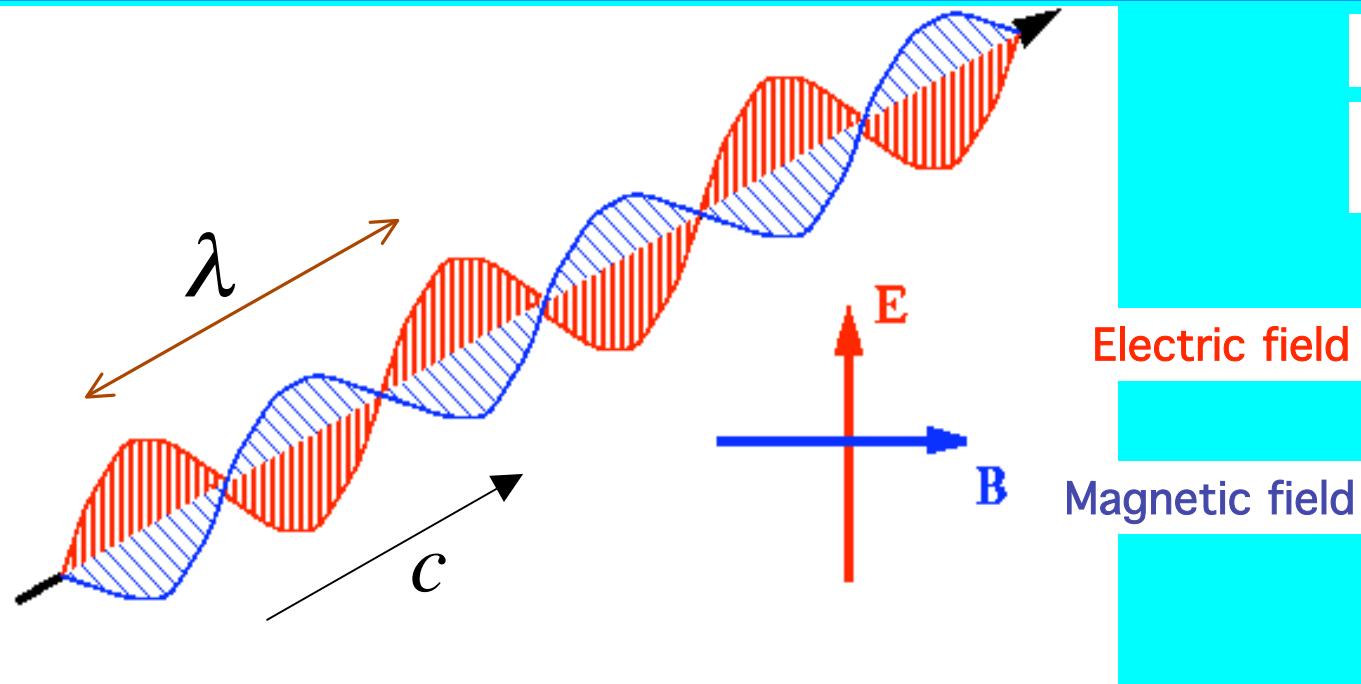
Würzburg (Germany)
November 8, 1895



Josef Albert repr.
Hand des Anatomen Geheimrath von Kölle in Würzburg.
Im Physikalischen Institut der Universität Würzburg
am 23. Januar 1896 mit X-Strahlen aufgenommen
von

Electromagnetic waves

Paolo
Fornasini
Univ. Trento



Speed (in vacuum):

$$c \approx 3 \times 10^8 \text{ m/s}$$

Electric field

Magnetic field

Wavelength

$$\lambda$$

$$1 \text{ \AA} = 10^{-10} \text{ m}$$

$$1 \text{ nm} = 10^{-9} \text{ m}$$

Frequency

$$\nu = c / \lambda$$

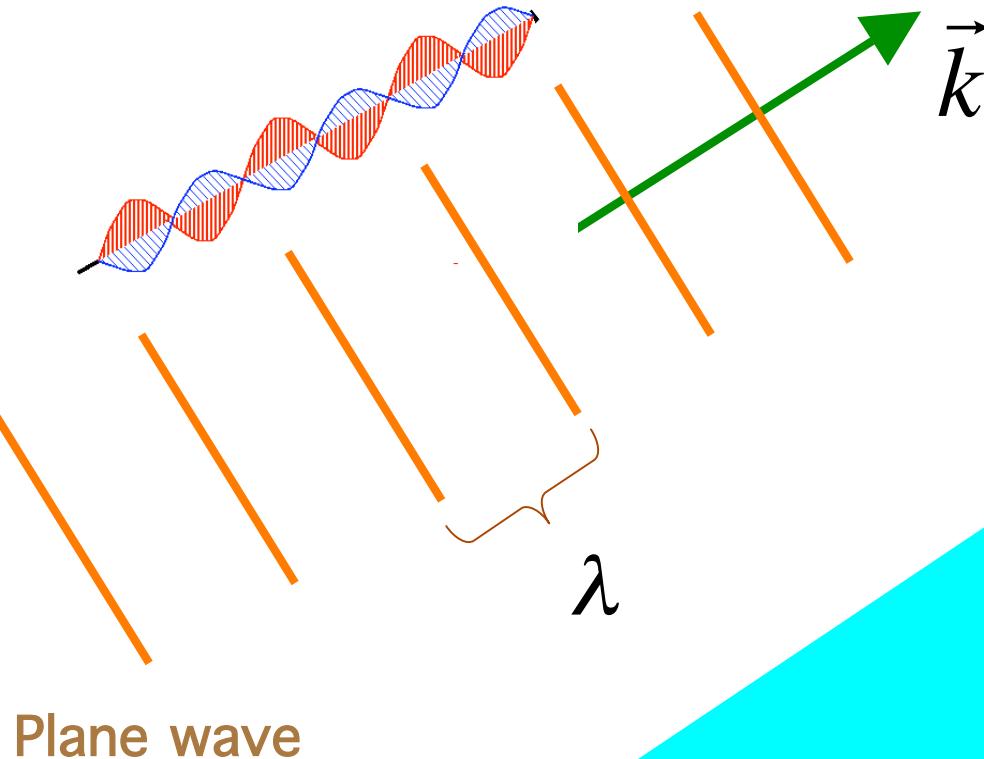
$$\omega = 2\pi\nu$$

Photon energy

$$E = h\nu = \hbar\omega$$

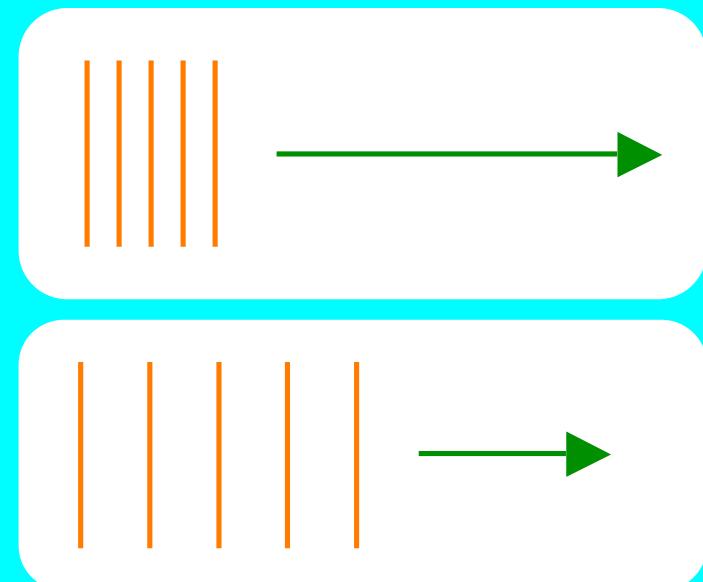
Wave-vector

Paolo
Fornasini
Univ. Trento



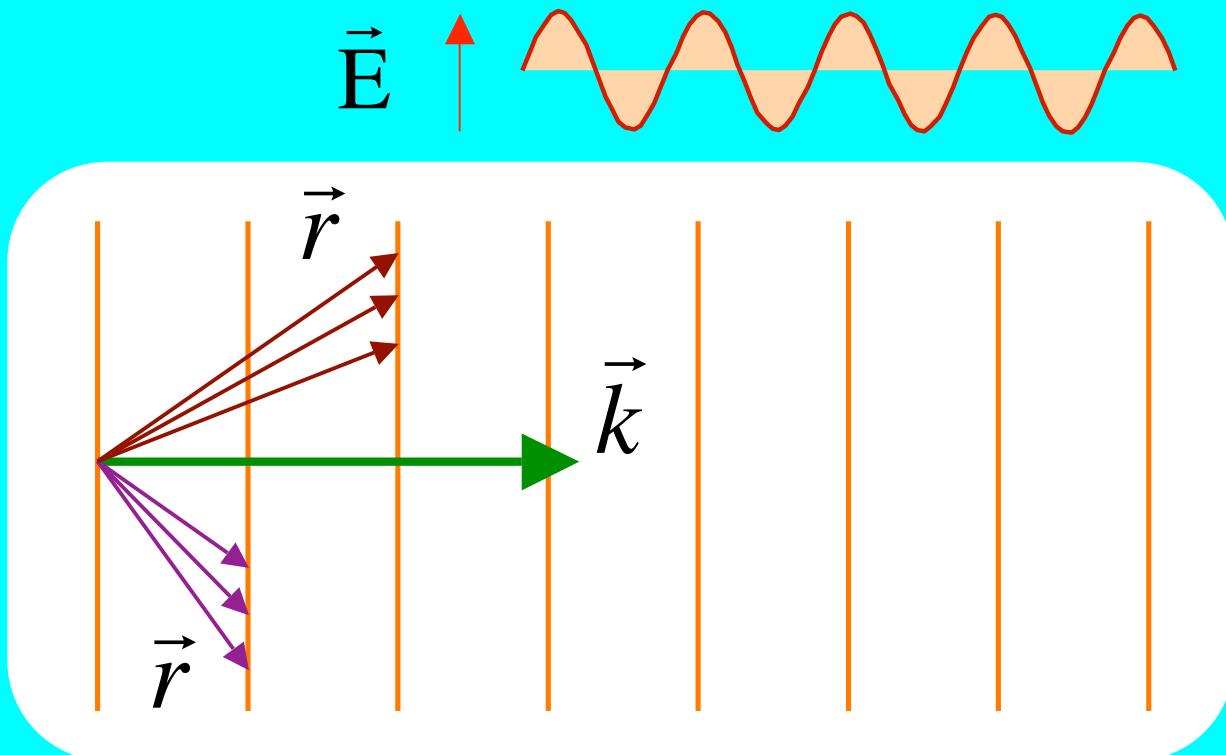
Plane wave

$$k = \frac{2\pi}{\lambda}$$



Plane wave

Paolo
Fornasini
Univ. Trento



$$k = \frac{2\pi}{\lambda}$$

$$\vec{E}(\vec{r}) = \vec{E}_0 \cos(\vec{k} \cdot \vec{r}) = \vec{E}_0 \quad \text{for} \quad \vec{k} \cdot \vec{r} = 2n\pi$$

$$\text{Re}\left\{ e^{i\vec{k} \cdot \vec{r}} \right\}$$

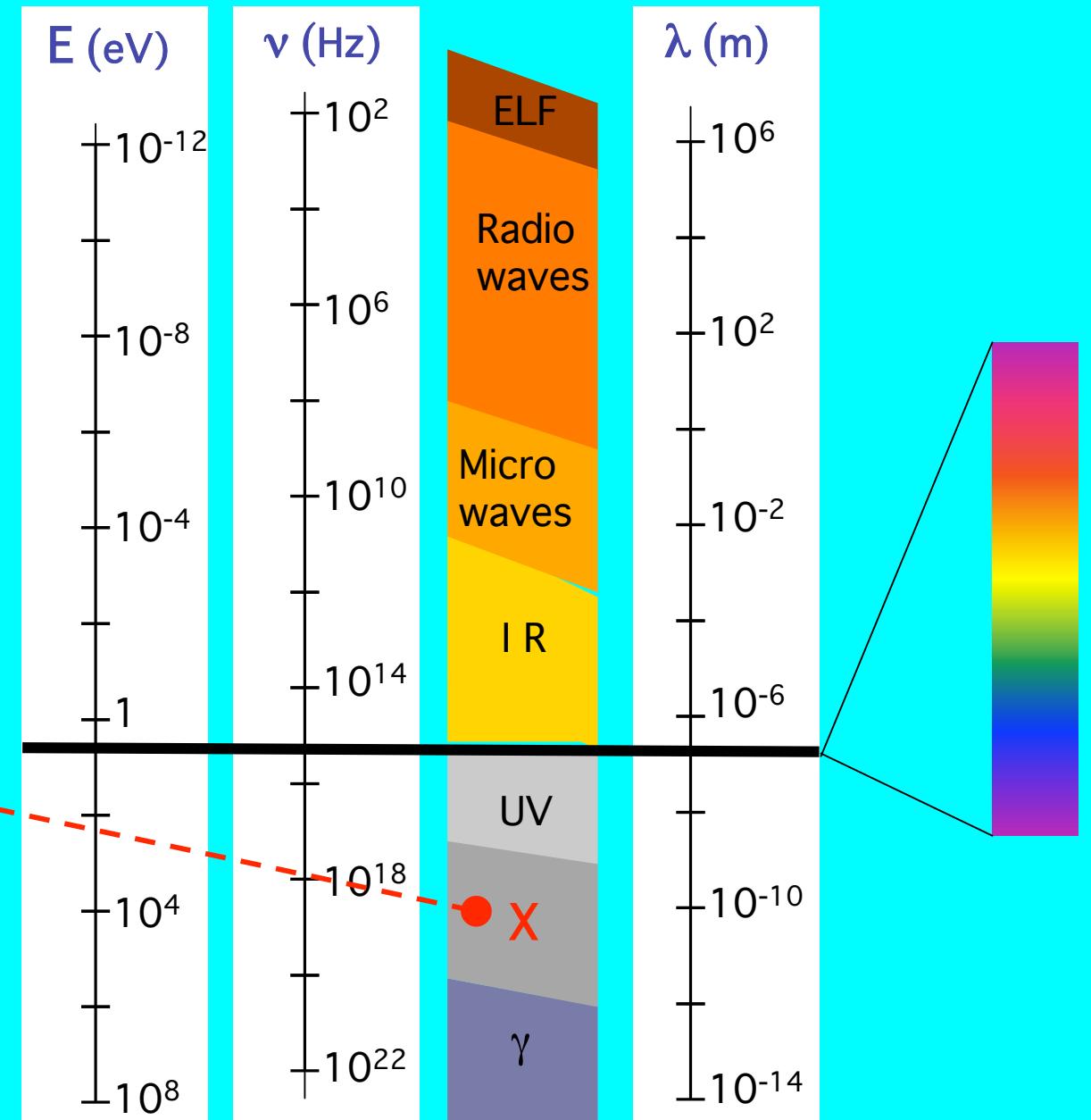
Complex
notation

Electromagnetic spectrum

Photon energy
Wavelength
 $E = h\nu = hc / \lambda$

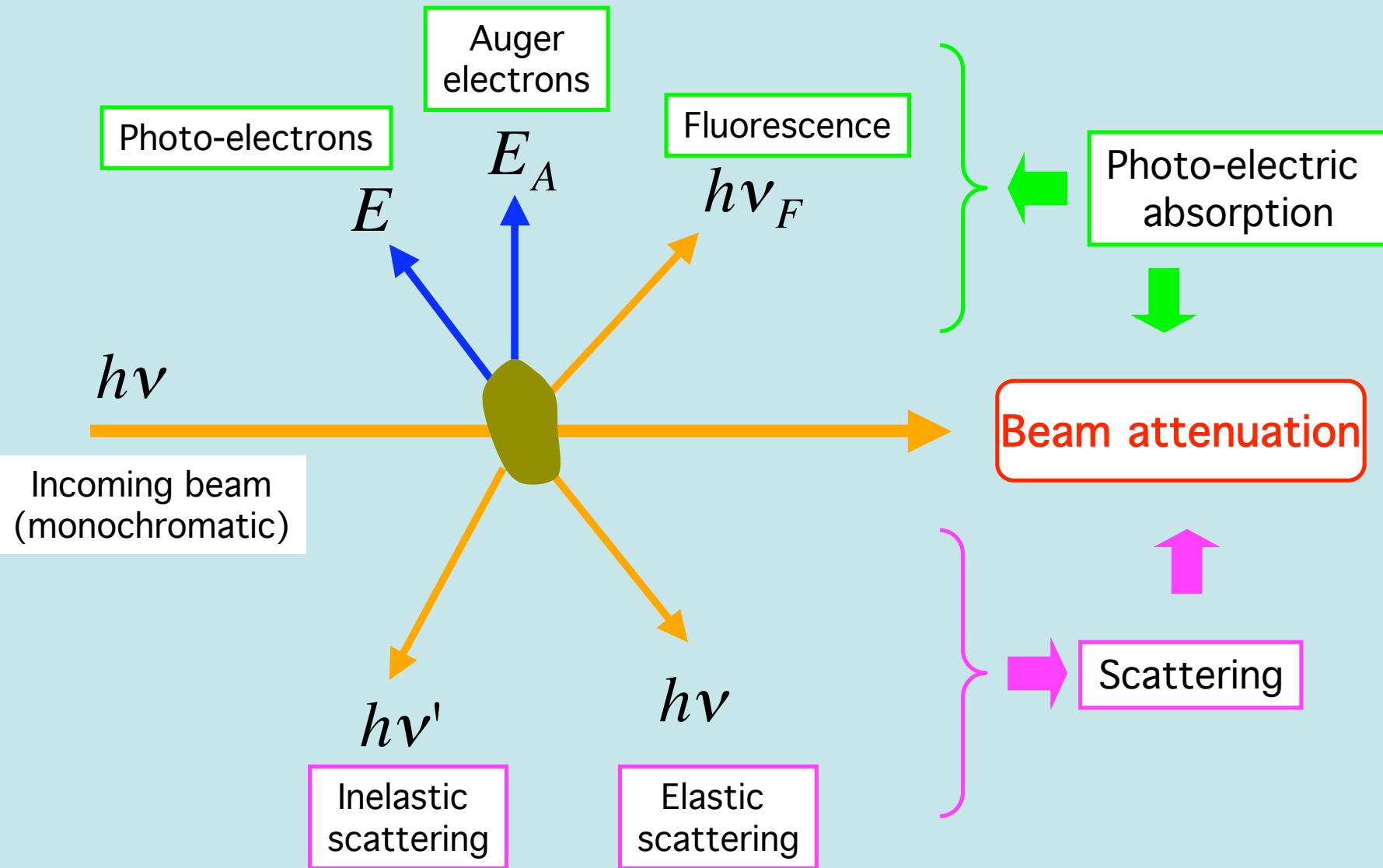
Frequency

X-rays
 $\lambda \approx 0.01 \div 10 \text{ \AA}$
 $\nu \approx 10^{17} \div 10^{20} \text{ Hz}$
 $E \approx 0.4 \div 400 \text{ keV}$



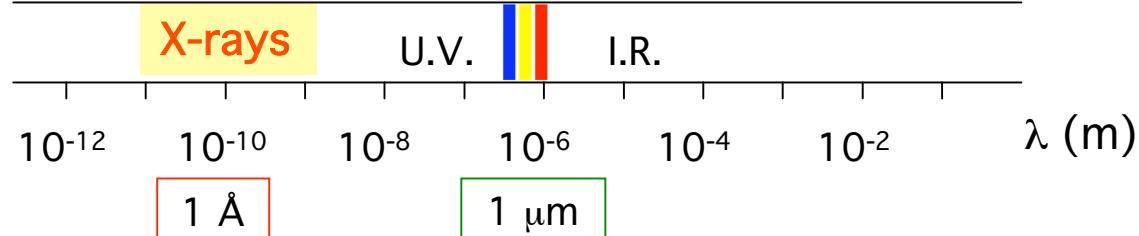
Interaction of x-rays with matter

Paolo
Fornasini
Univ. Trento



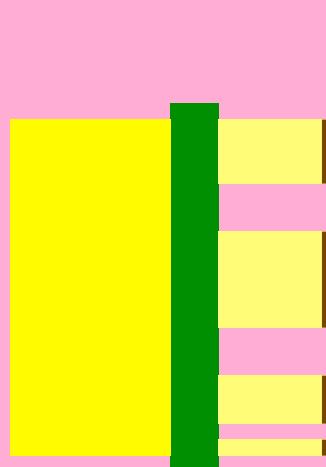
X-RAYS and X-ray techniques

Paolo
Fornasini
Univ. Trento

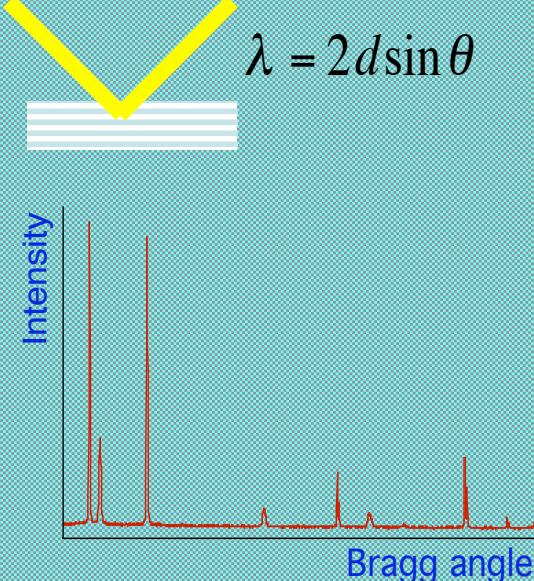


$$E[\text{keV}] = \frac{12.4}{\lambda[\text{\AA}]}$$

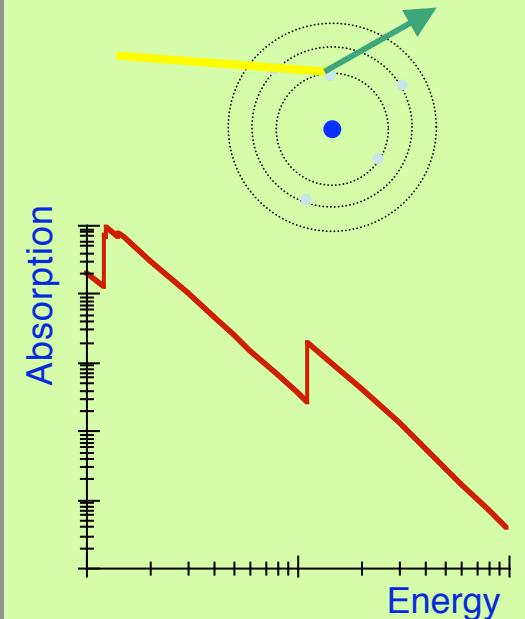
Imaging



Scattering



Spectroscopy





Crystals

Crystals

Paolo
Fornasini
Univ. Trento

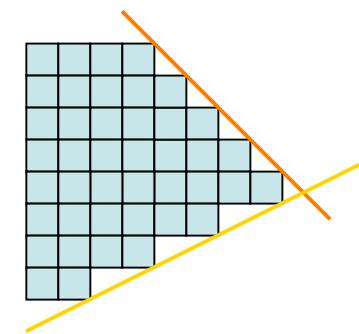


Quartz crystal (SiO_2)

Macroscopic regularities
(e.g. constancy of angles)



Classification of crystals



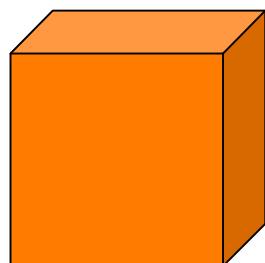
Regular packing
of microscopic structural units
R.J. Haüy (1743-1822)

Atoms and crystals

Paolo
Fornasini
Univ. Trento

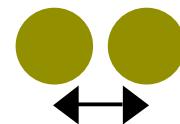
HYPOTHESIS: Structural units = atoms

Example: NaCl



Atomic masses: Na 38.12×10^{-24} g
Cl 58.85×10^{-24} g

Cubic structure
 $1 \text{ cm}^3 \text{ m} = 2.165 \text{ g}$
 $N = 44.6 \times 10^{21} \text{ atoms}$



$0.28 \text{ nm} = 2.8 \text{ \AA}$

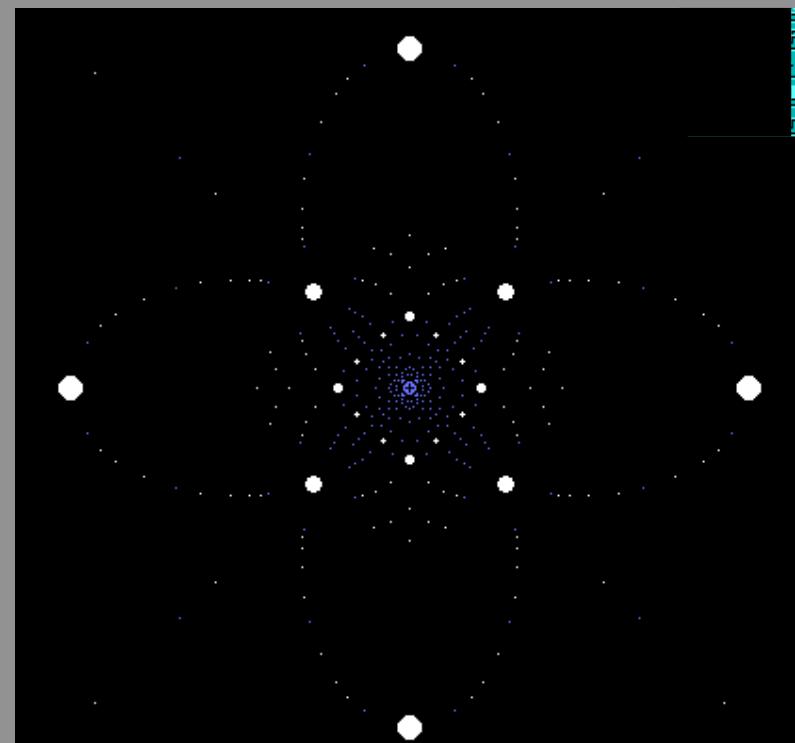
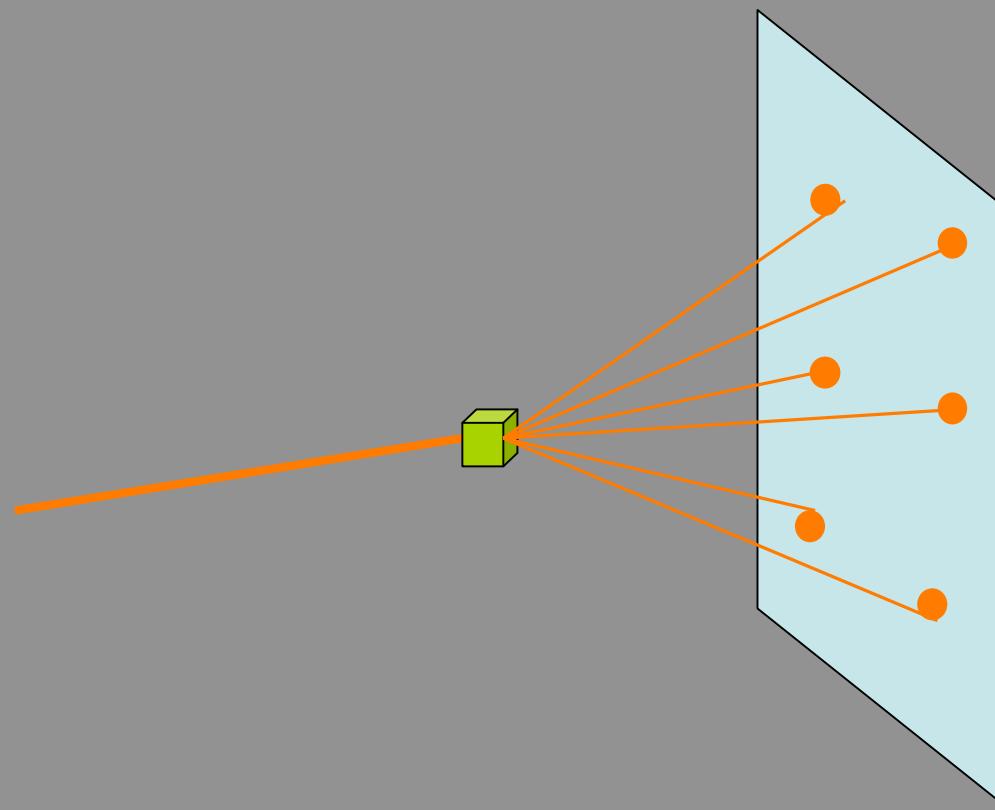
CONCLUSION:

Inter-atomic distances
Atomic dimensions

\approx X-ray wavelengths

X-ray diffraction from crystals

Paolo
Fornasini
Univ. Trento



Munich, 1912:

- Max von Laue
- W. Friedrich & P.Knipping

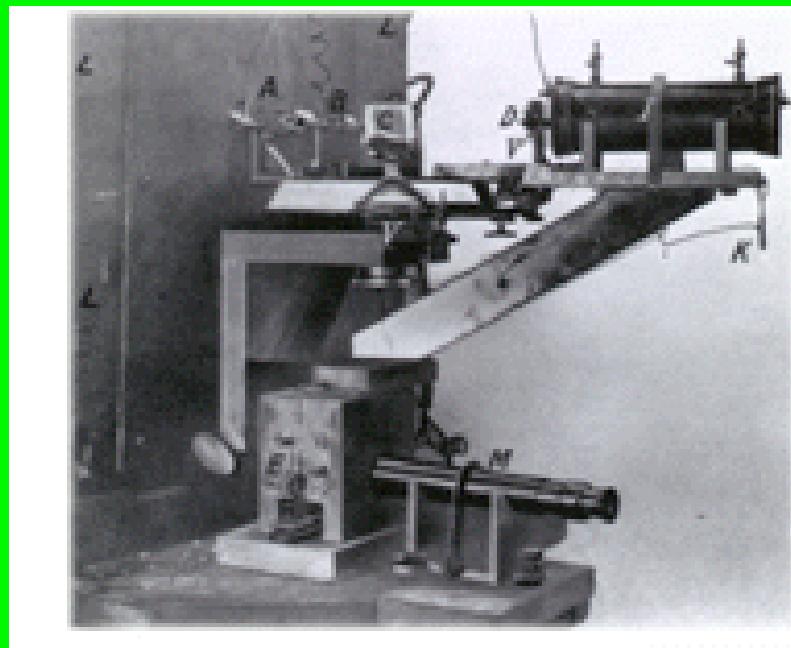
Crystallography

Paolo
Fornasini
Univ. Trento



William Henry Bragg
(1862-1942)

Cambridge, 1912/13



Bragg spectrometer



William Lawrence Bragg
(1890-1971)

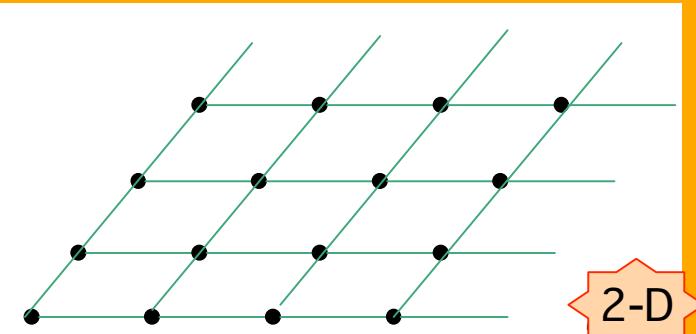
Crystal structure

Paolo
Fornasini
Univ. Trento

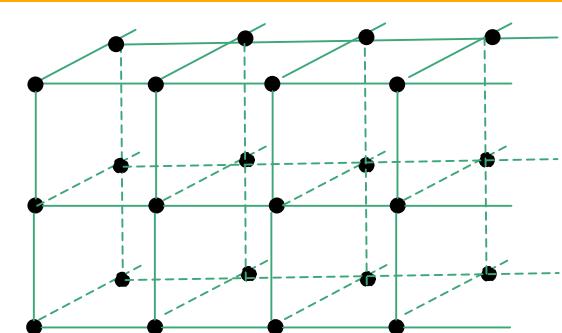
Bravais lattice



1-D



2-D



3-D

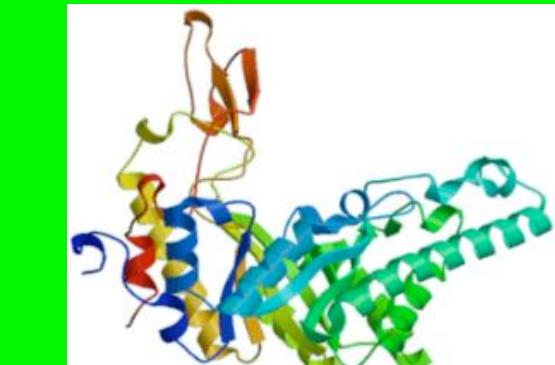
+

Basis

Atom



Molecule



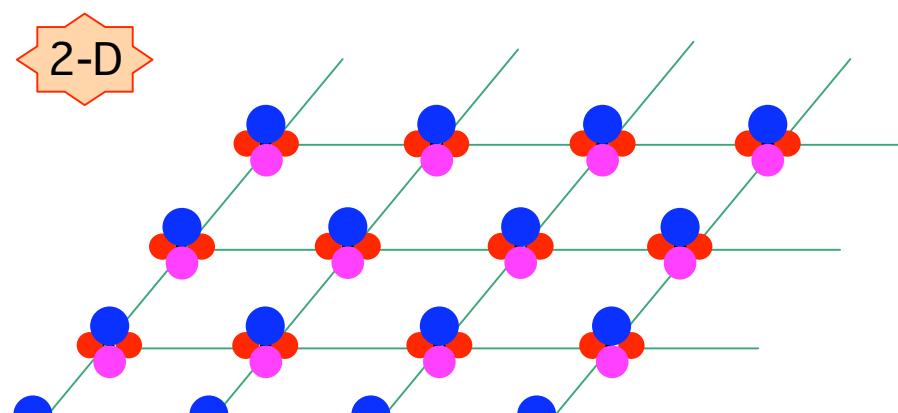
Protein

Bravais lattice + basis

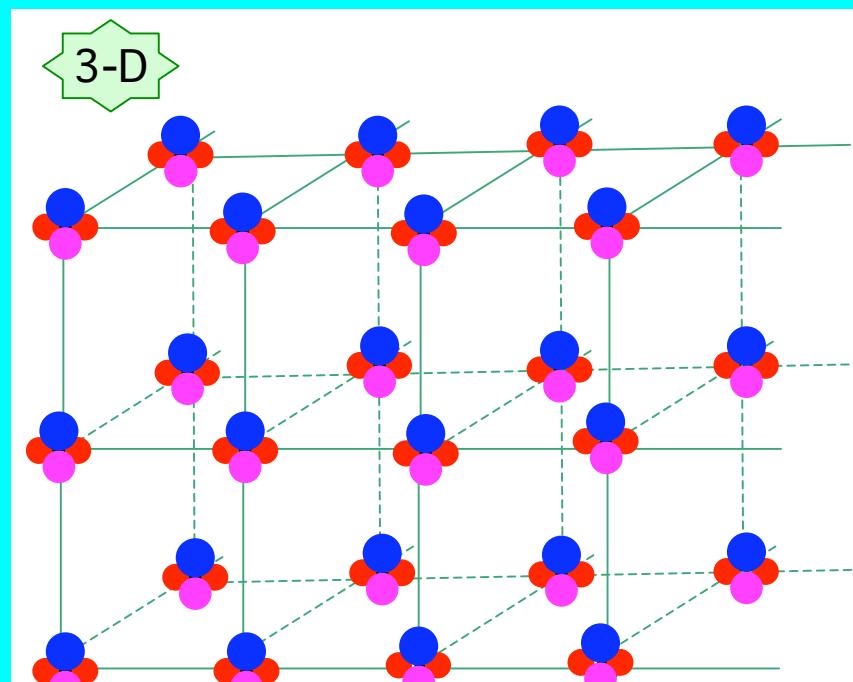
Paolo
Fornasini
Univ. Trento



1-D



2-D

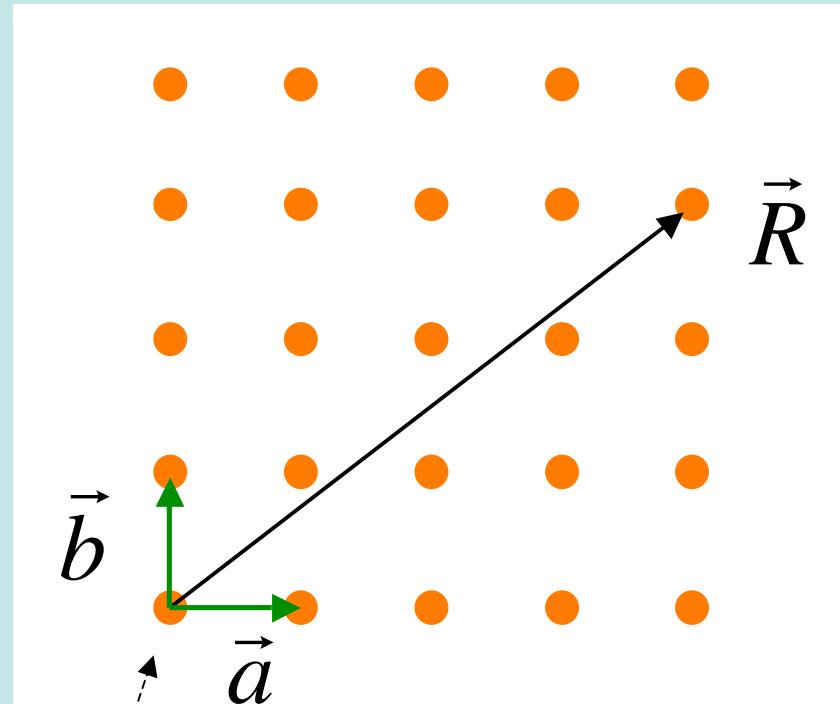




Crystal lattices

Translation vectors (2D)

Paolo
Fornasini
Univ. Trento



Arbitrary origin

2-D

For every lattice point

$$\vec{R} = n_1 \vec{a} + n_2 \vec{b}$$

integers

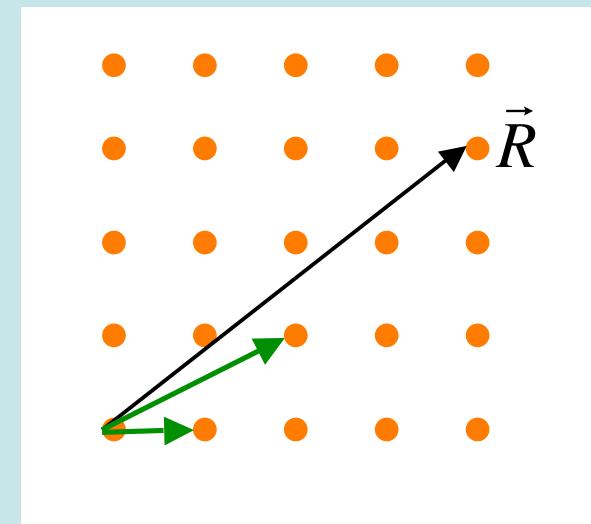
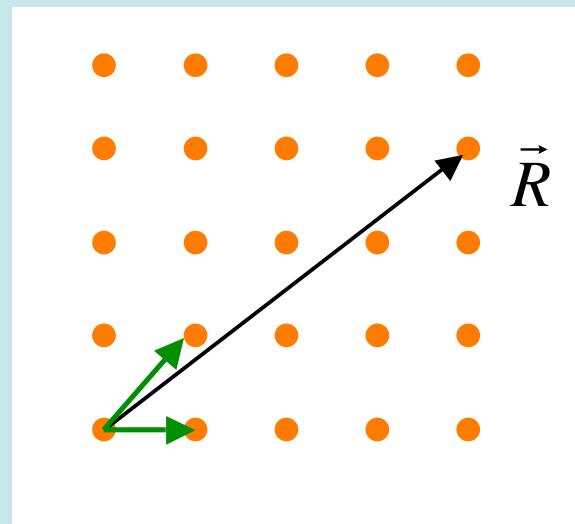
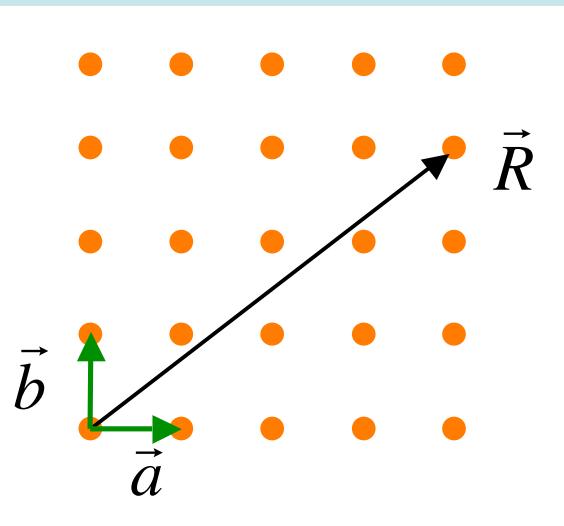
primitive
vectors

Primitive vectors (2D)

Paolo
Fornasini
Univ. Trento

2-D

$$\vec{R} = n_1 \vec{a} + n_2 \vec{b}$$

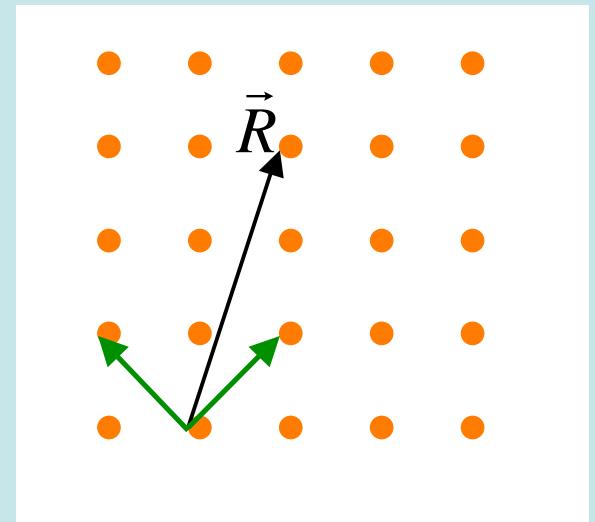
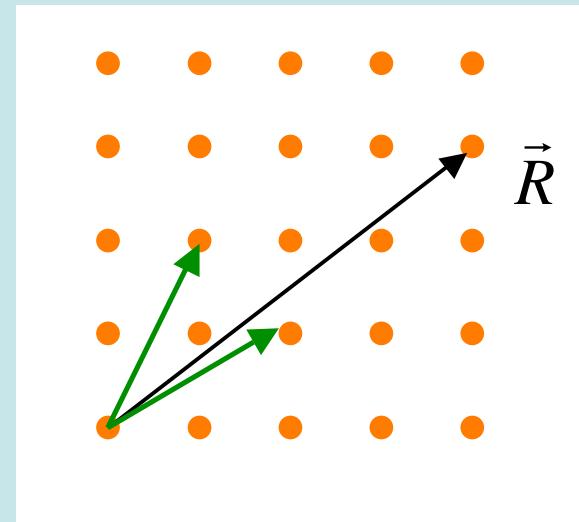
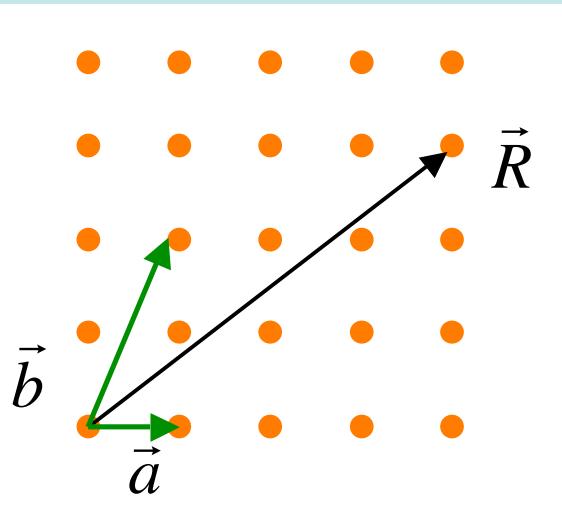


Different choices of primitive vectors \vec{a}, \vec{b}

Non-primitive vectors (2D)

2-D

Not all \vec{a}, \vec{b} pairs are primitive

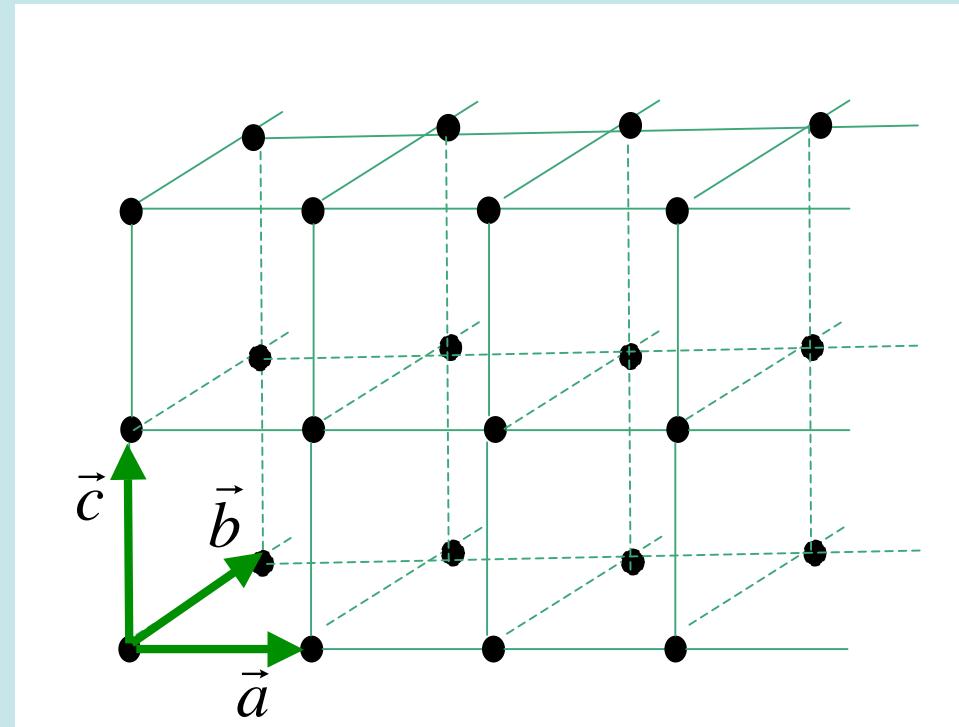


$$\vec{R} \neq n_1 \vec{a} + n_2 \vec{b}$$

Primitive vectors (3D)

$$\vec{R} = n_1 \vec{a} + n_2 \vec{b} + n_3 \vec{c}$$

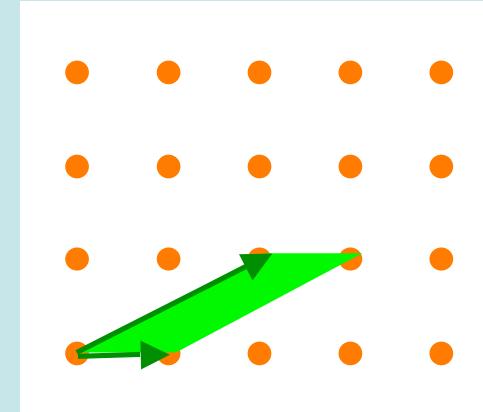
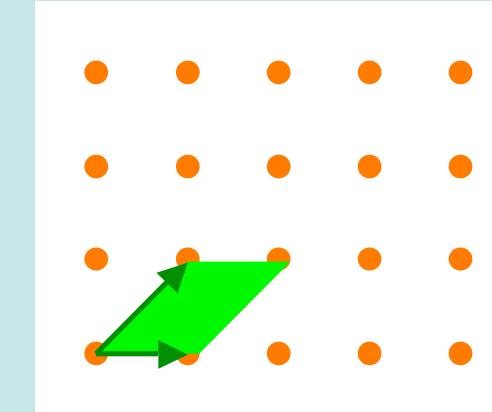
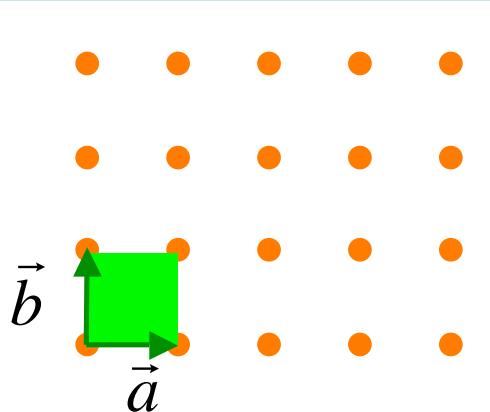
3-D



Different choices of primitive vectors $\vec{a}, \vec{b}, \vec{c}$

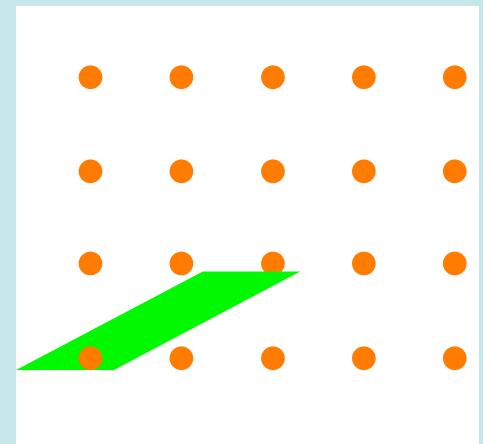
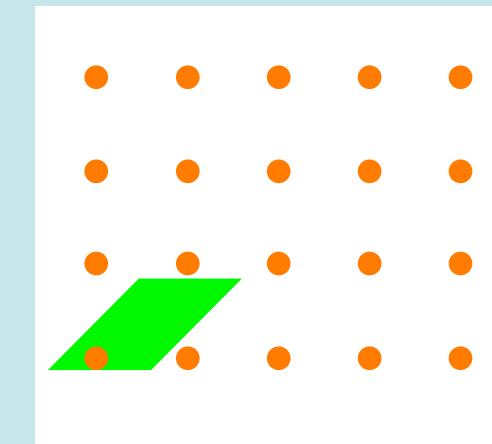
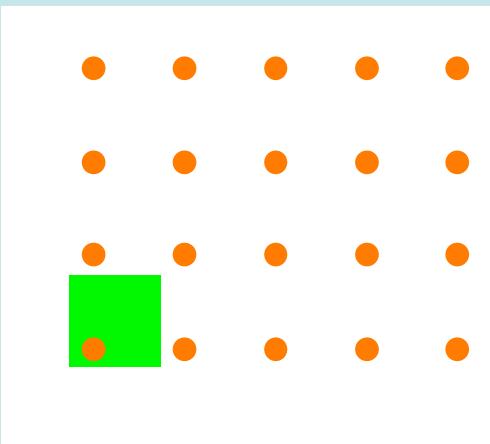
Primitive unit cells (2D)

Paolo
Fornasini
Univ. Trento



2-D

Different choices of primitive unit cells

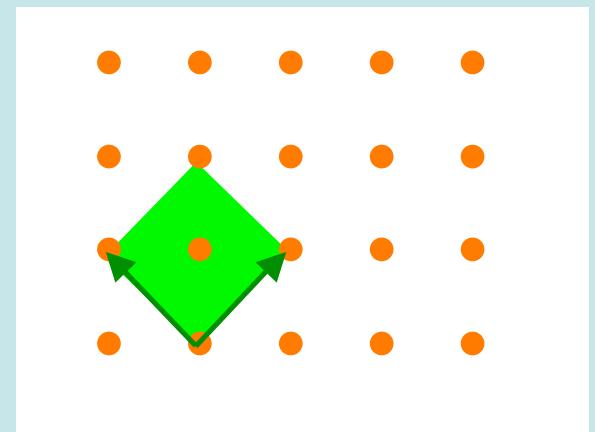
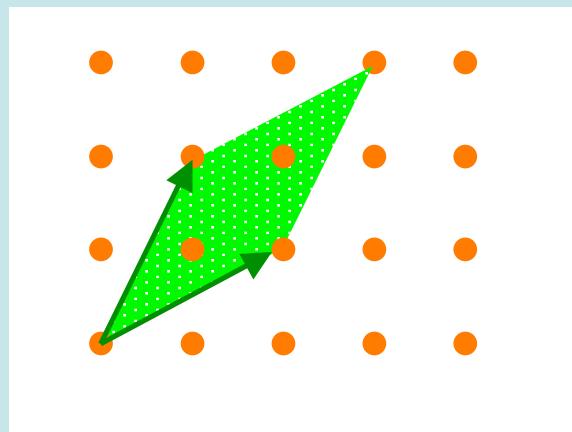
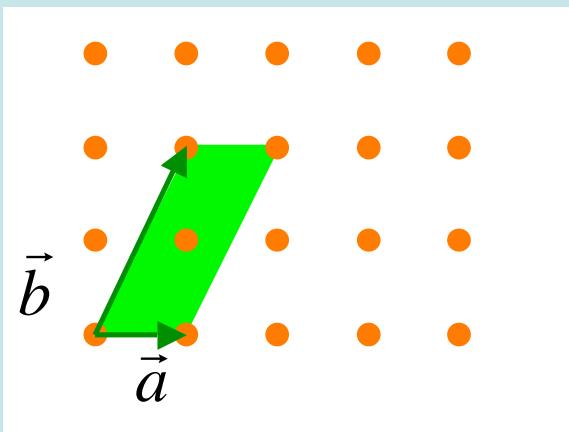


Primitive cell = 1 lattice point

Conventional unit cells (2D)

Paolo
Fornasini
Univ. Trento

2-D

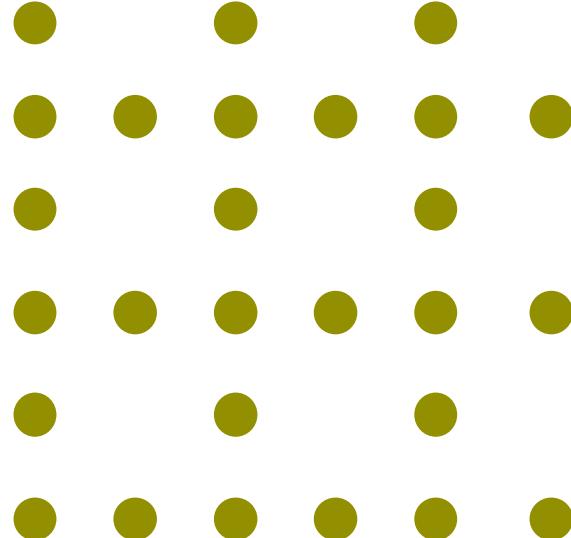


More than 1 lattice point per unit cell

Non-Bravais lattices

Paolo
Fornasini
Univ. Trento

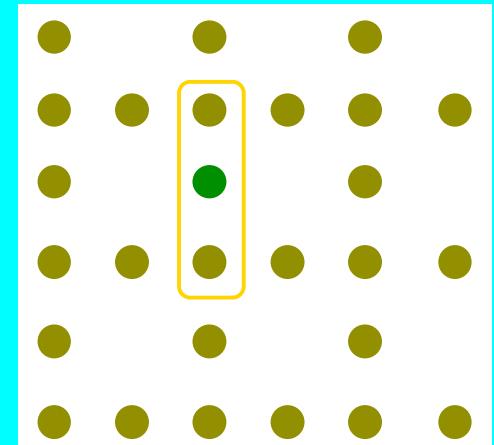
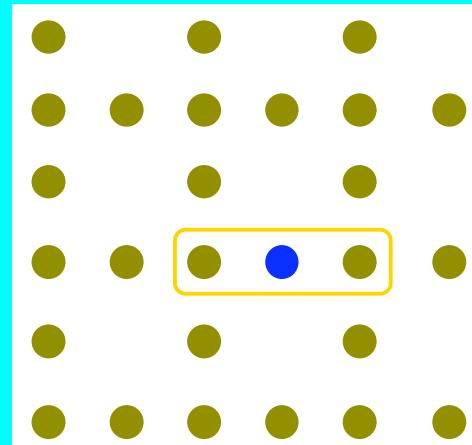
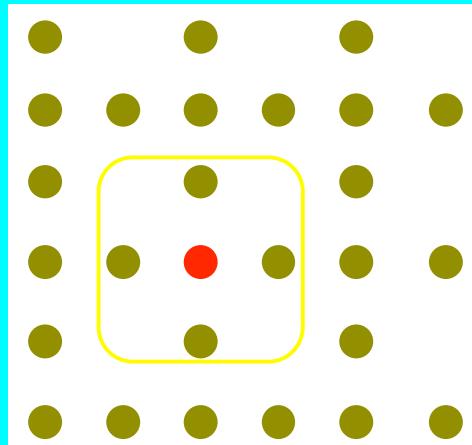
Atoms



2-D

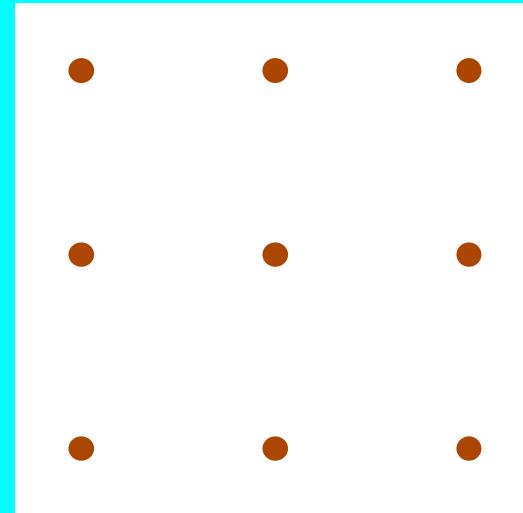
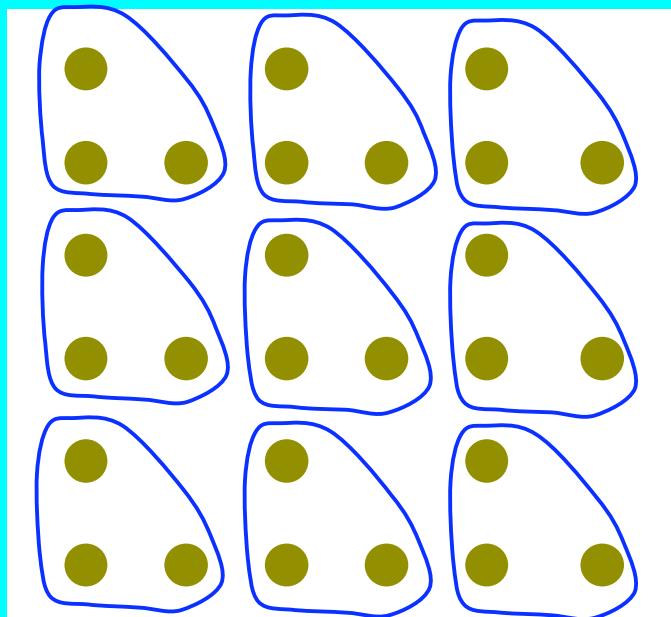
$$\vec{R} \neq n_1 \vec{a} + n_2 \vec{b}$$

Un-equivalent
sites

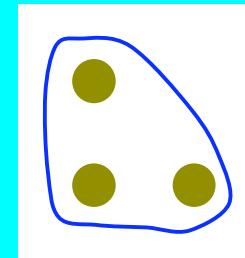


Bravais lattices

Paolo
Fornasini
Univ. Trento



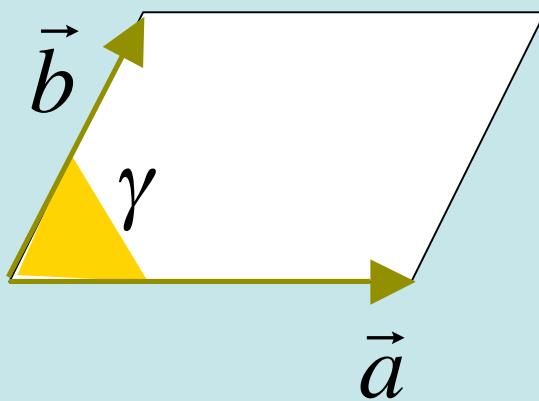
Bravais
lattice



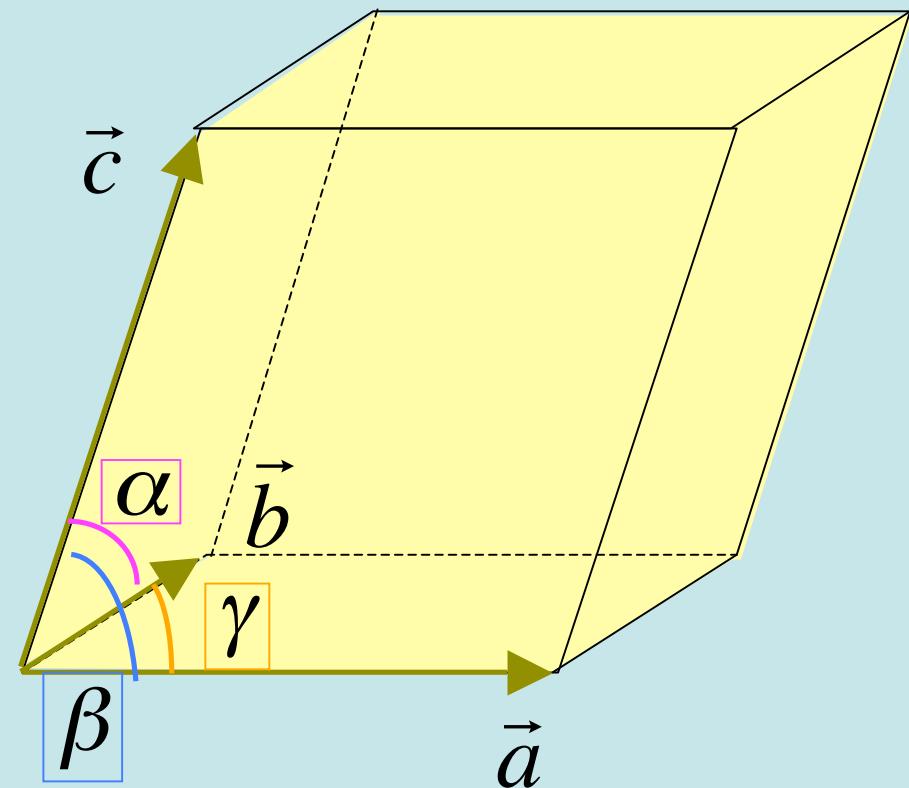
Basis

Classification of cells

2-D



3-D



$a \quad b \quad c$

latin

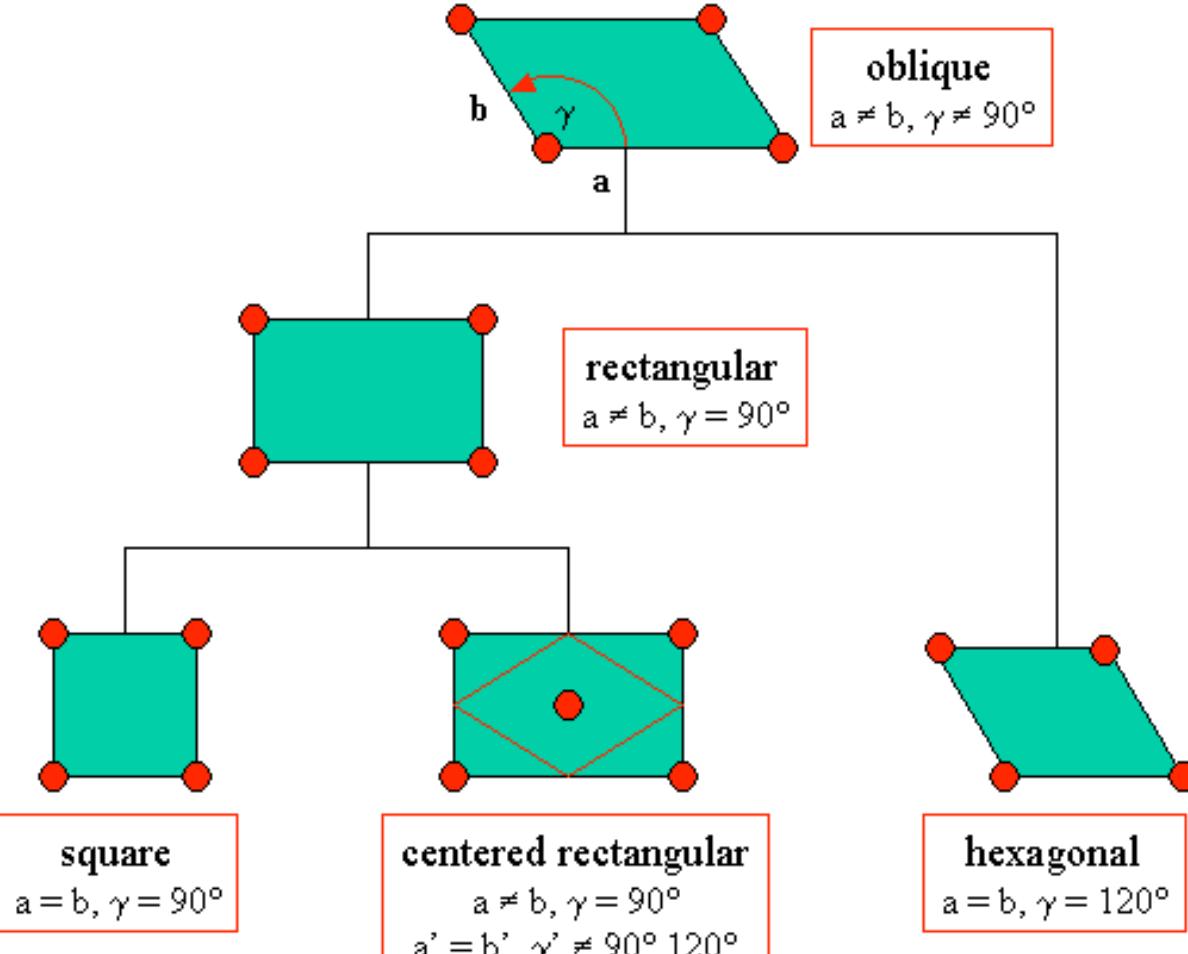
$\alpha \quad \beta \quad \gamma$

greek

Surface Bravais lattice

Paolo
Fornasini
Univ. Trento

2-D

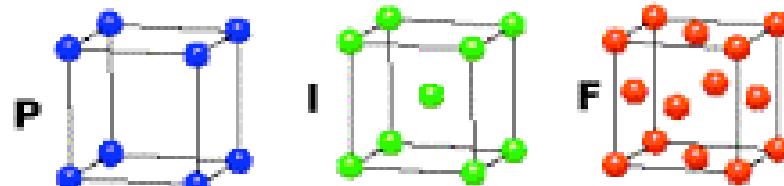


3-D Bravais lattices

Paolo
Fornasini
Univ. Trento

CUBIC

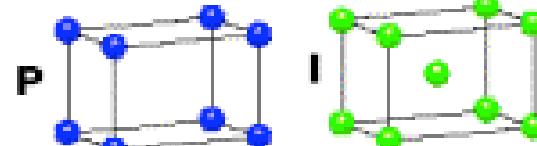
$a = b = c$
 $\alpha = \beta = \gamma = 90^\circ$



3-D

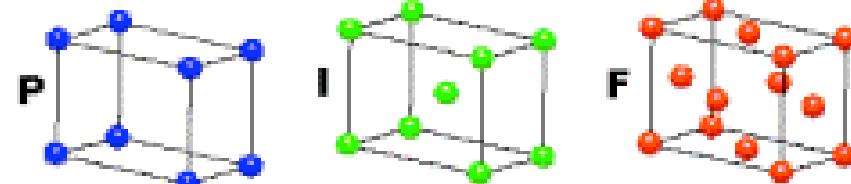
TETRAGONAL

$a = b \neq c$
 $\alpha = \beta = \gamma = 90^\circ$



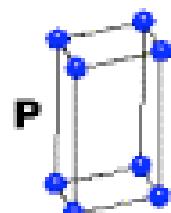
ORTHORHOMBIC

$a \neq b \neq c$
 $\alpha = \beta = \gamma = 90^\circ$



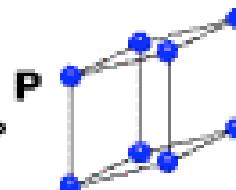
HEXAGONAL

$a = b \neq c$
 $\alpha = \beta = 90^\circ$
 $\gamma = 120^\circ$



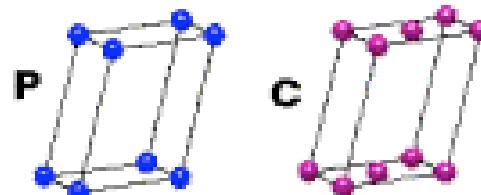
TRIGONAL

$a = b = c$
 $\alpha = \beta = \gamma \neq 90^\circ$



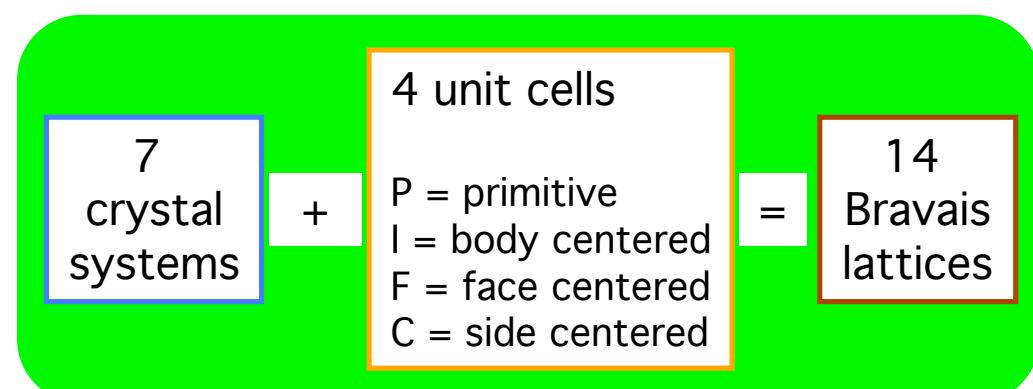
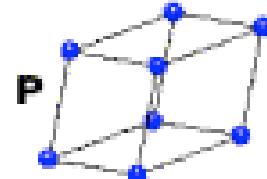
MONOCLINIC

$a \neq b \neq c$
 $\alpha = \gamma = 90^\circ$
 $\beta \neq 120^\circ$



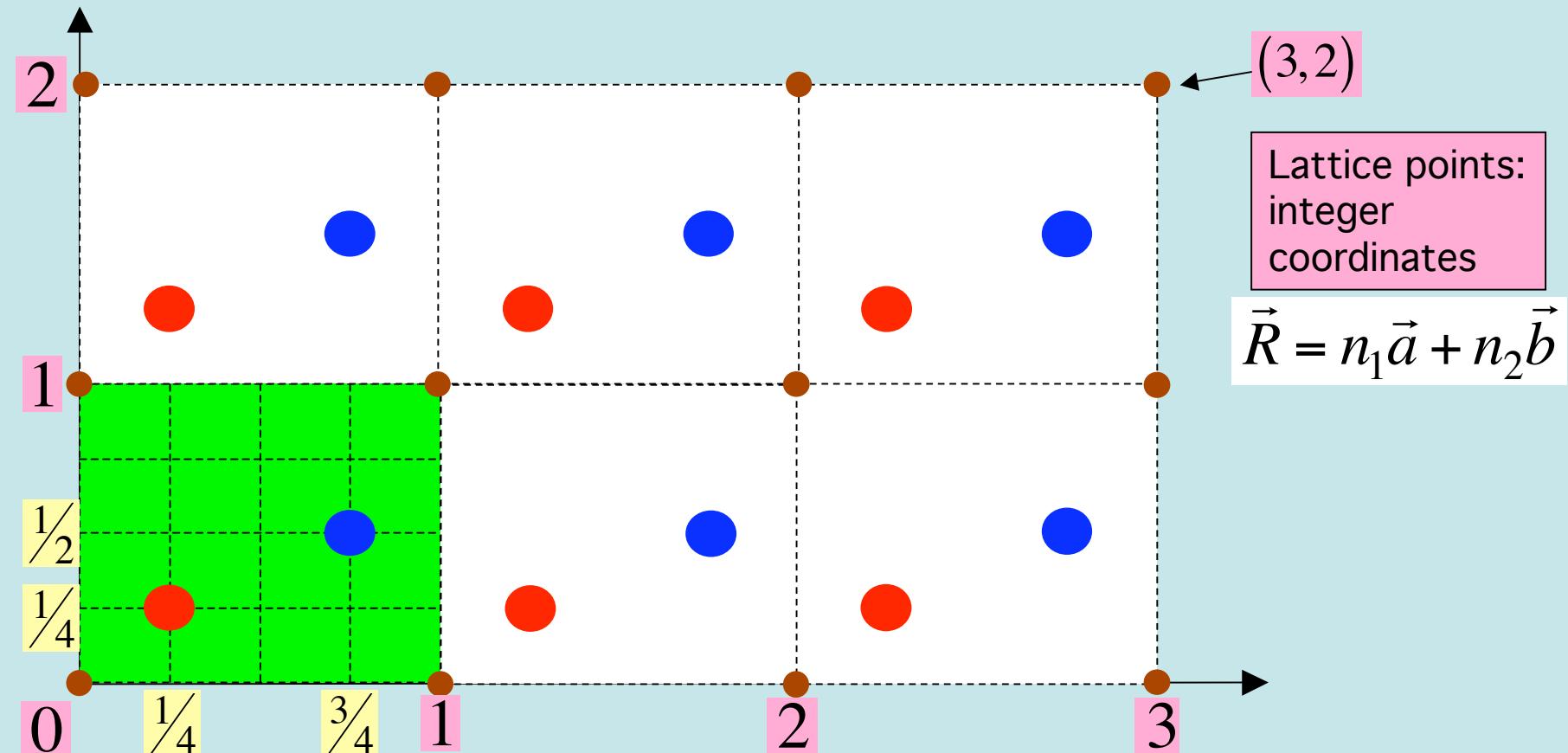
TRICLINIC

$a \neq b \neq c$
 $\alpha \neq \beta \neq \gamma \neq 90^\circ$



Coordinates

Paolo
Fornasini
Univ. Trento



Inside cell: fractional coordinates

$$\bullet \quad \left(\frac{1}{4}, \frac{1}{4} \right)$$

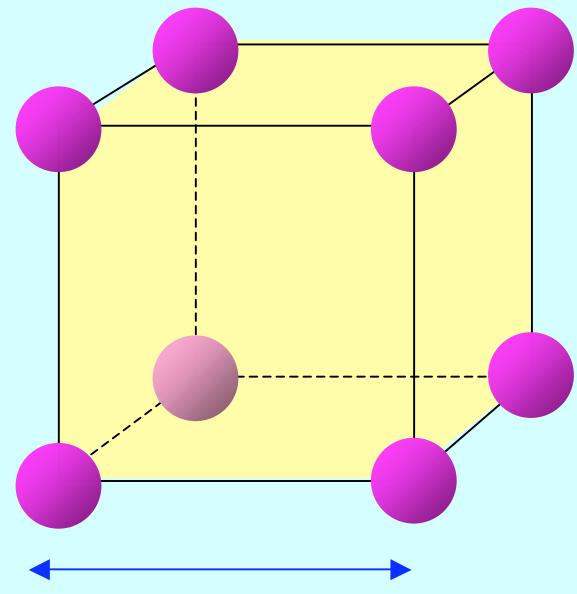
$$\bullet \quad \left(\frac{3}{4}, \frac{1}{2} \right)$$



Some relevant crystal structures

Simple cubic lattice

Paolo
Fornasini
Univ. Trento

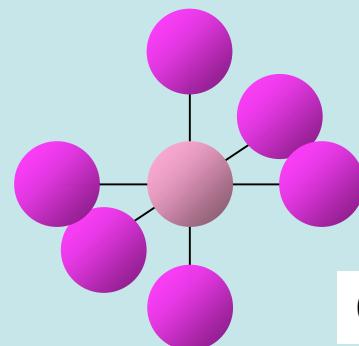
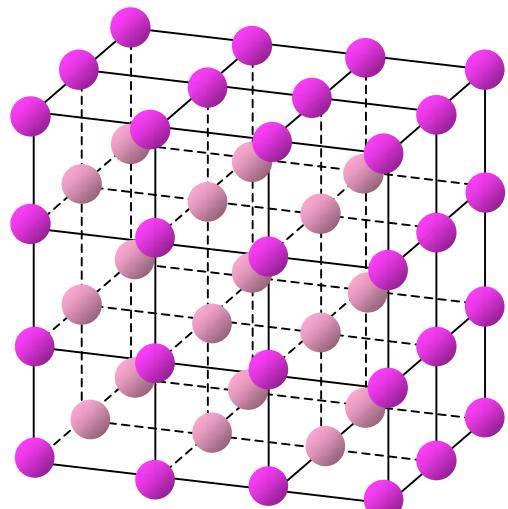


lattice parameter

Primitive unit cell
(1 lattice point per cell)

84-Po $a=3.35 \text{ \AA}$

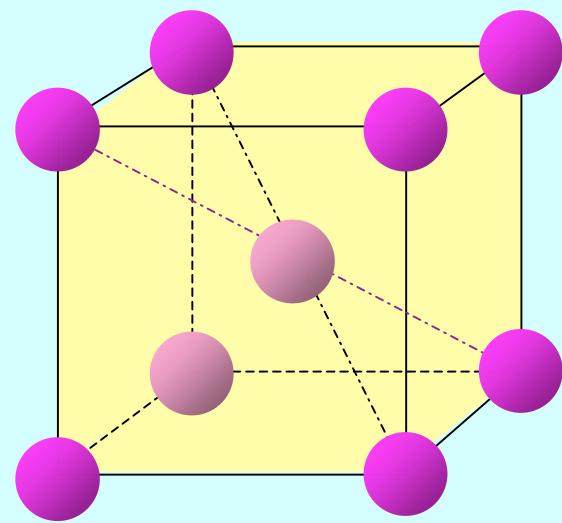
Bravais lattice



Coordination number = 6

Body centered cubic lattice (bcc)

Paolo
Fornasini
Univ. Trento

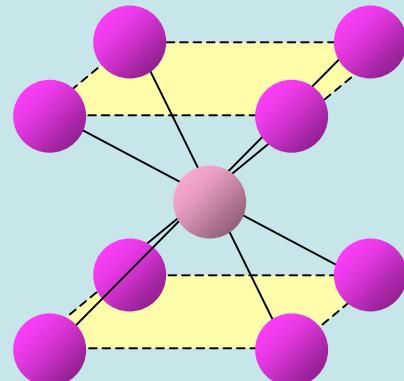


lattice parameter

conventional unit cell
(2 lattice points per cell)

24-Cr $a=2.88 \text{ \AA}$
26-Fe $a=2.87 \text{ \AA}$
42-Mo $a=3.15 \text{ \AA}$

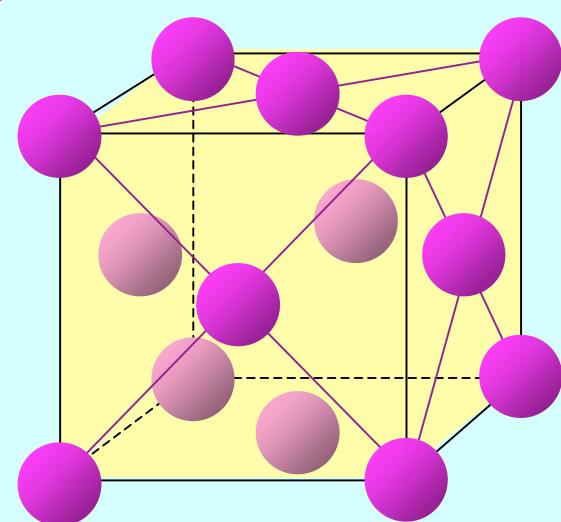
Bravais lattice



Coordination number = 8

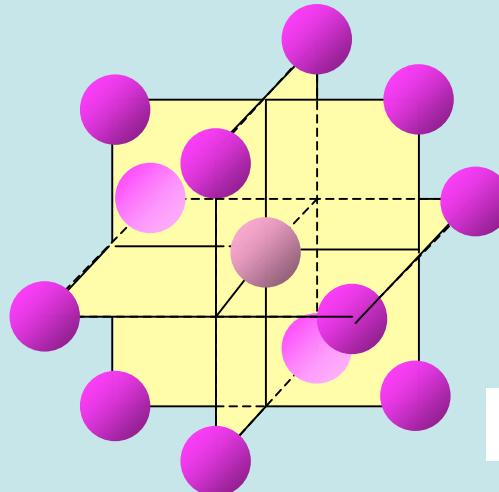
Face centered cubic lattice (fcc)

Paolo
Fornasini
Univ. Trento



lattice parameter

conventional unit cell
(4 lattice points per cell)

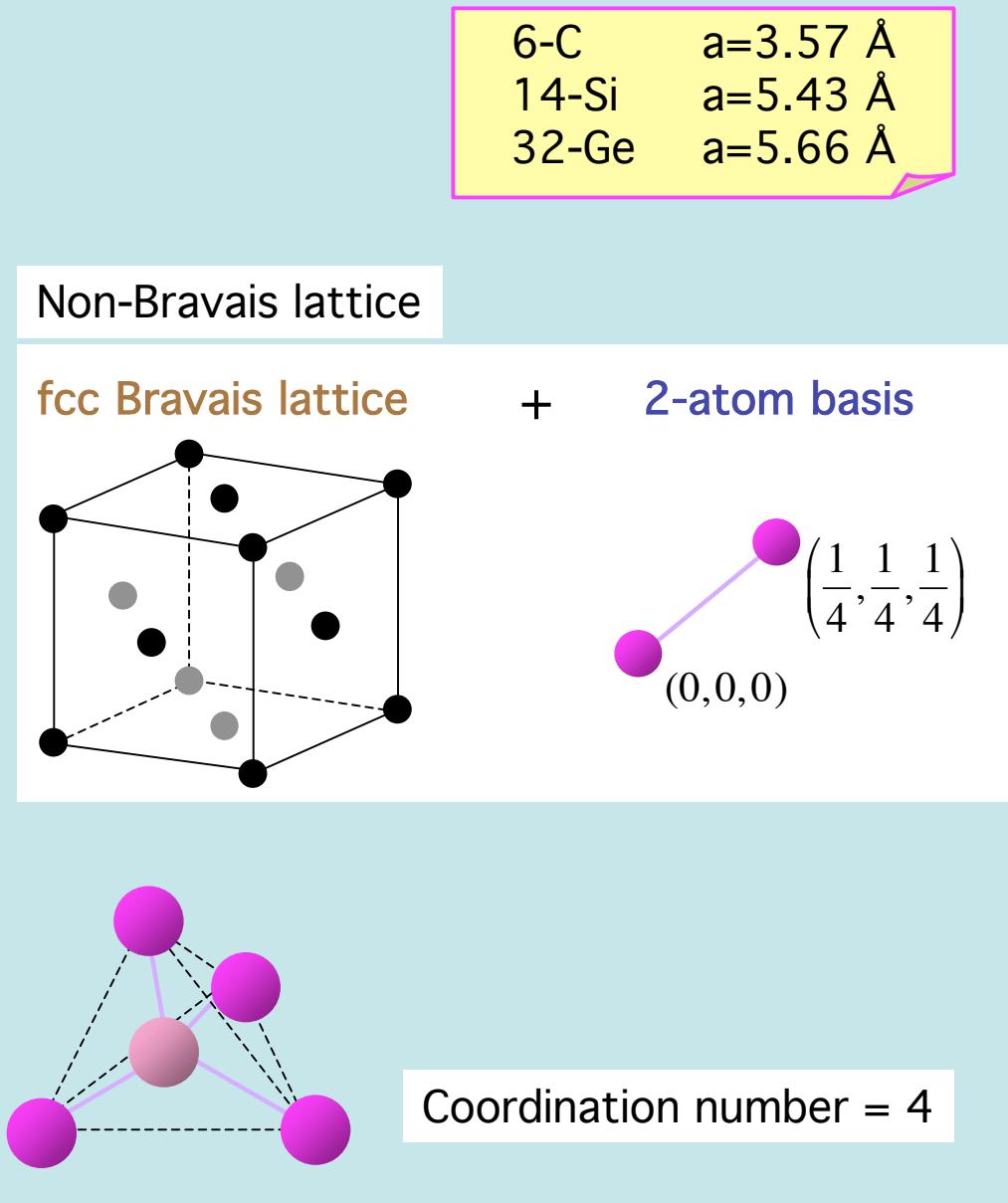
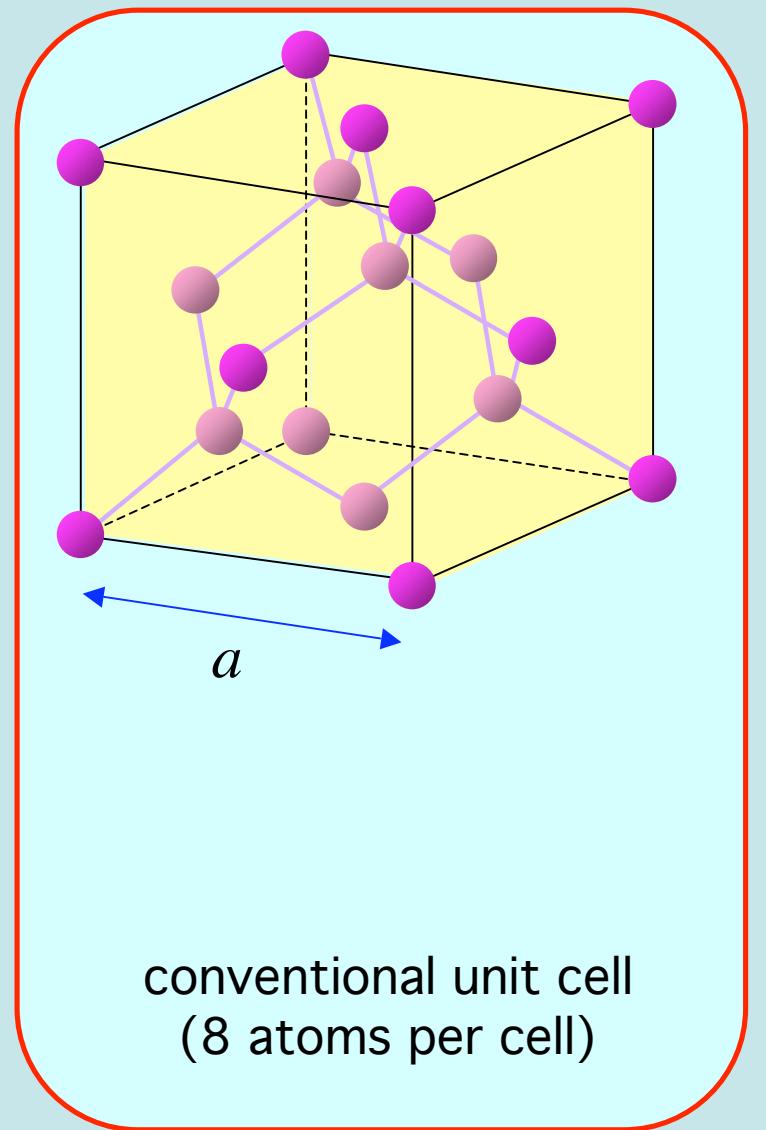


Coordination number = 12

29-Cu $a=3.61 \text{ \AA}$
47-Ag $a=4.09 \text{ \AA}$
79-Au $a=4.08 \text{ \AA}$

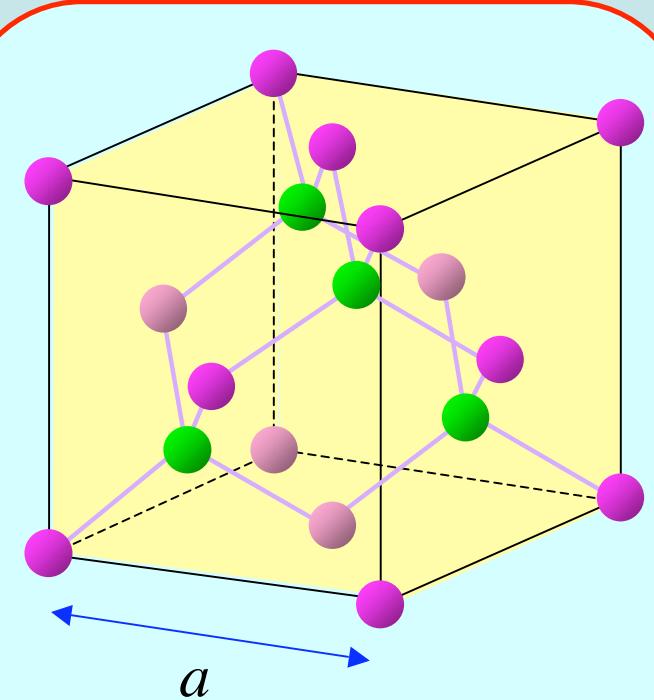
Diamond structure

Paolo
Fornasini
Univ. Trento



Zincblende (sphalerite) structure

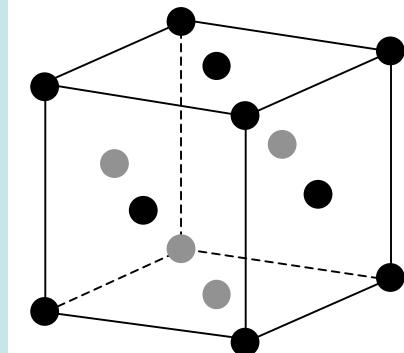
Paolo
Fornasini
Univ. Trento



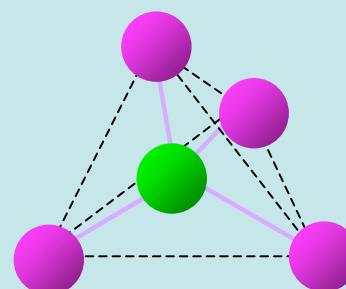
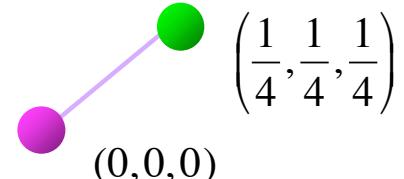
ZnS $a=5.41 \text{ \AA}$
GaAs $a=5.65 \text{ \AA}$
SiC $a=4.35 \text{ \AA}$

Non-Bravais lattice

fcc Bravais lattice



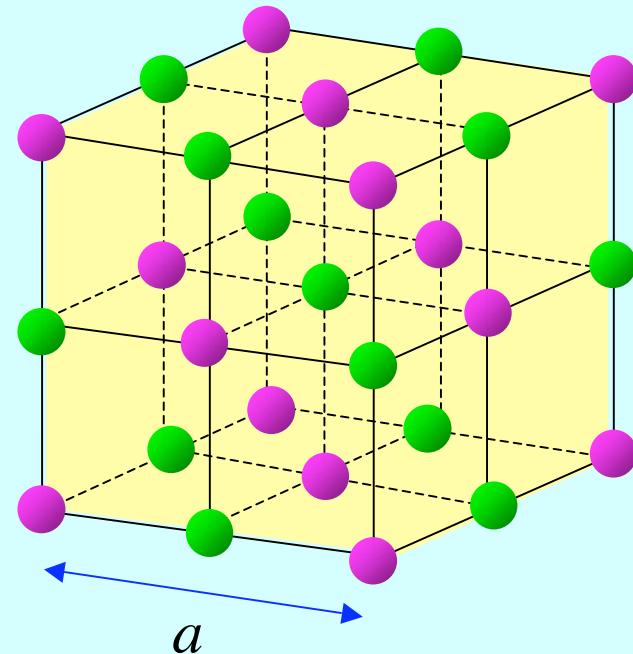
+ 2-atom basis



Coordination number = 4

Rock-salt (NaCl) structure

Paolo
Fornasini
Univ. Trento

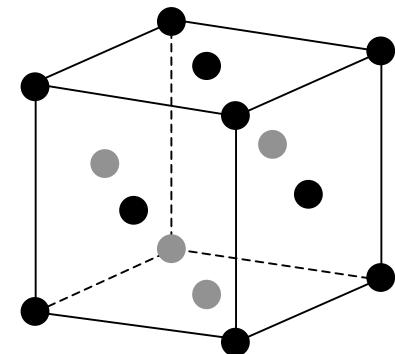


conventional unit cell
(8 atoms per cell)

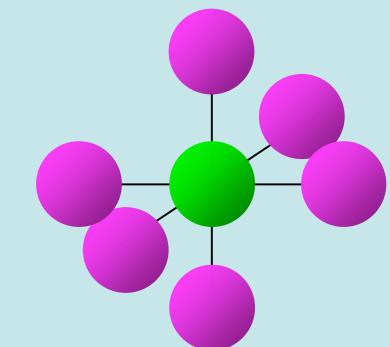
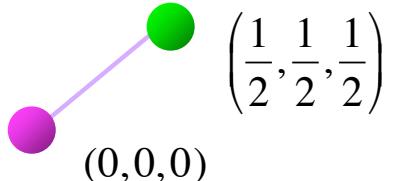
NaCl $a=5.64 \text{ \AA}$
KBr $a=6.60 \text{ \AA}$
CaO $a=4.81 \text{ \AA}$

Non-Bravais lattice

fcc Bravais lattice



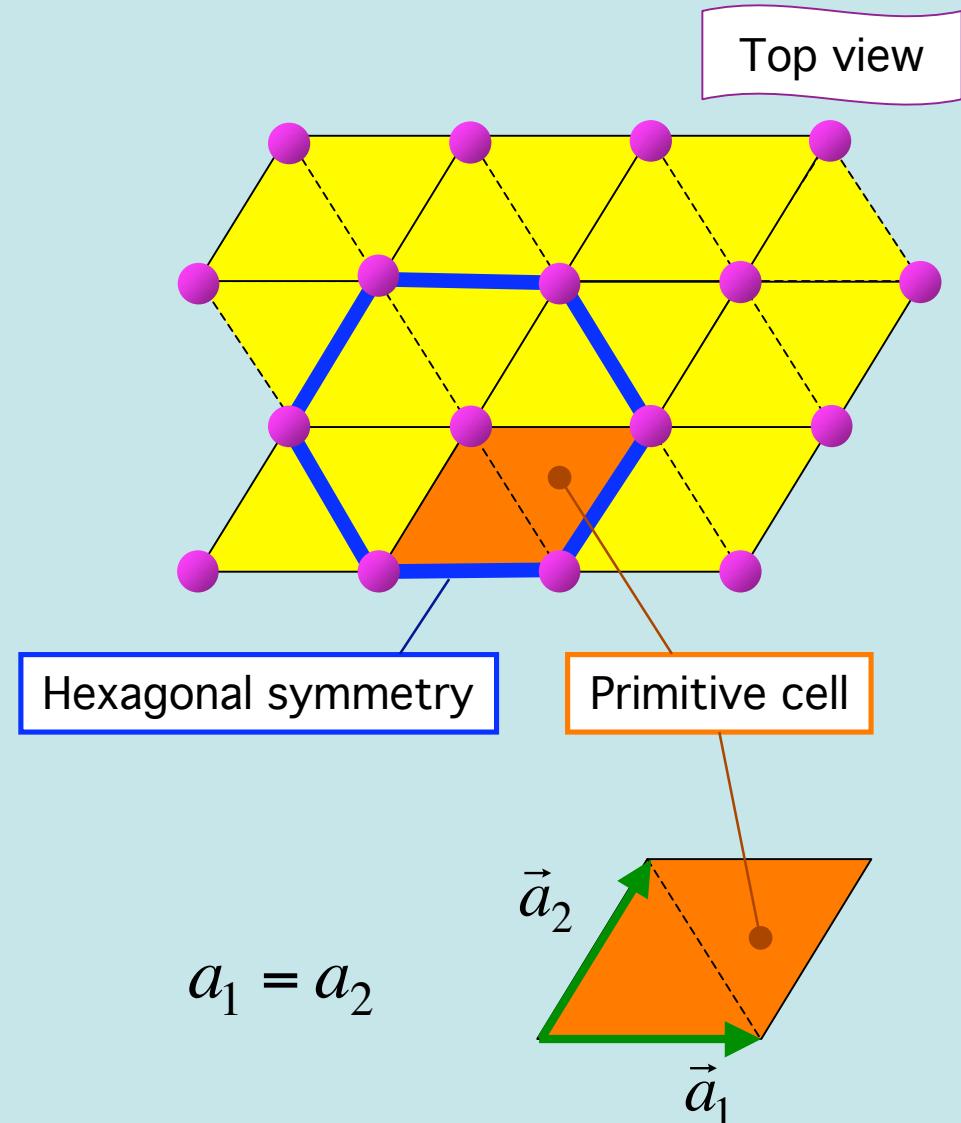
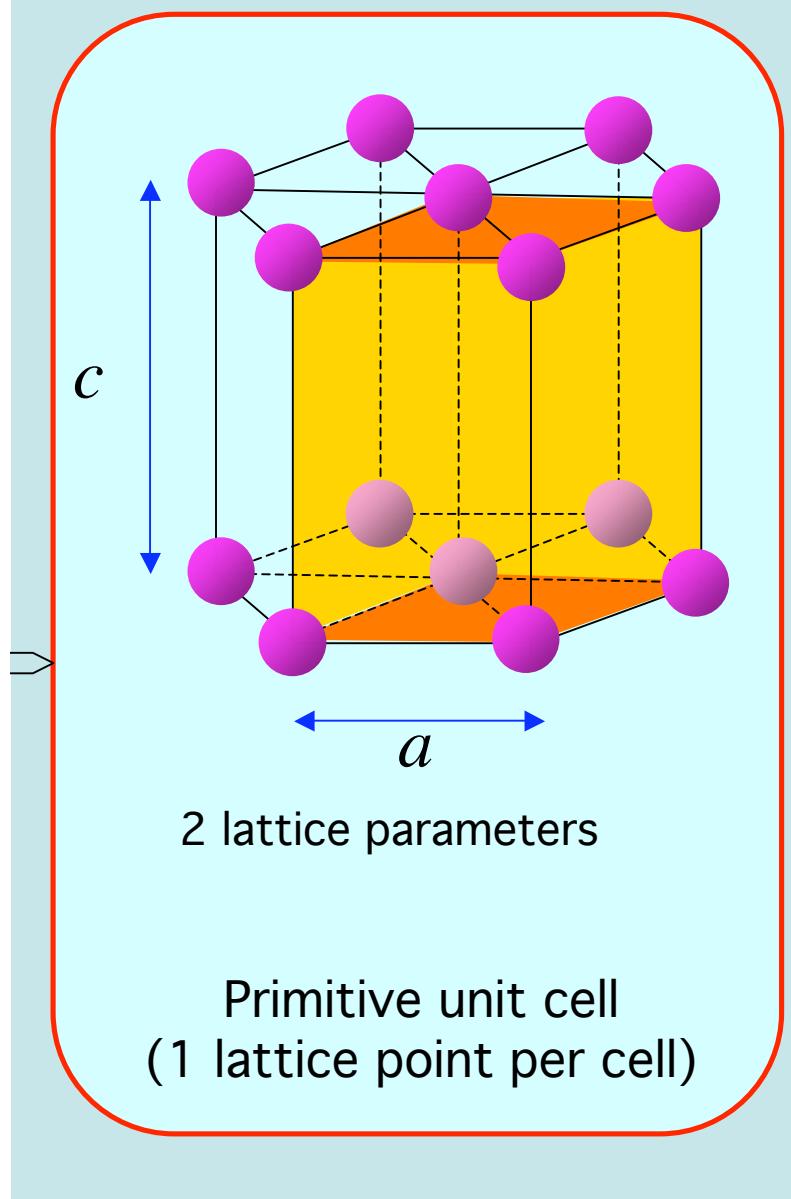
+ 2-atom basis



Coordination number = 6

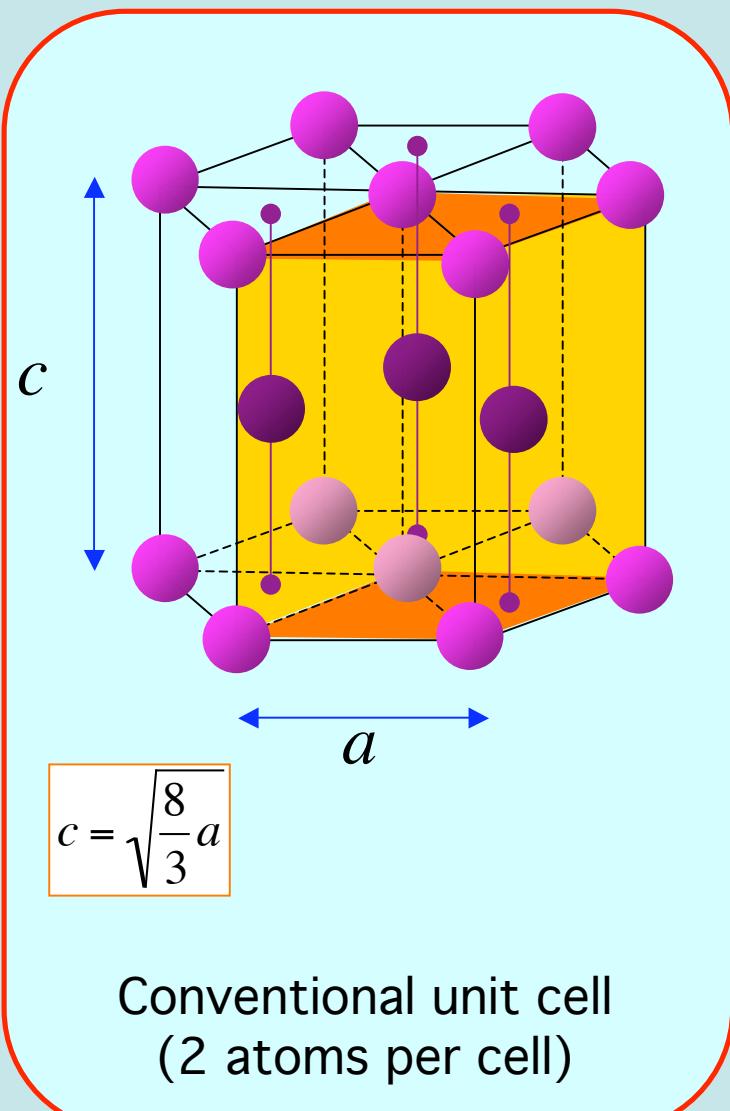
Simple hexagonal structure

Paolo
Fornasini
Univ. Trento



Hexagonal close packed structure

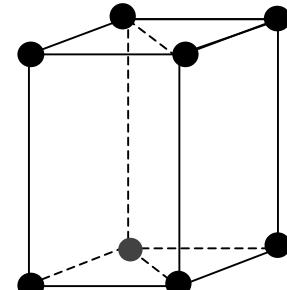
Paolo
Fornasini
Univ. Trento



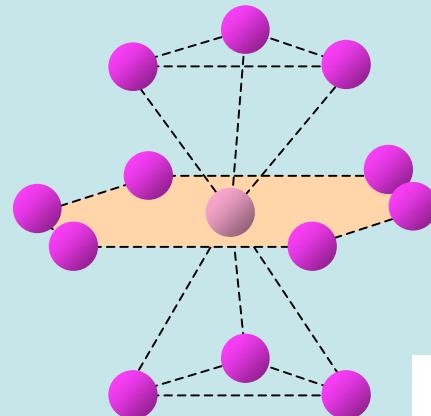
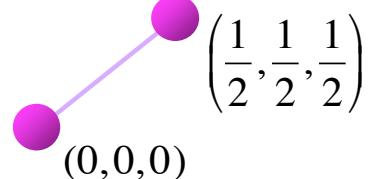
4-Be $a=2.29 \text{ \AA}$
12-Mg $a=3.21 \text{ \AA}$
48-Cd $a=2.98 \text{ \AA}$

Non-Bravais lattice

primitive cell



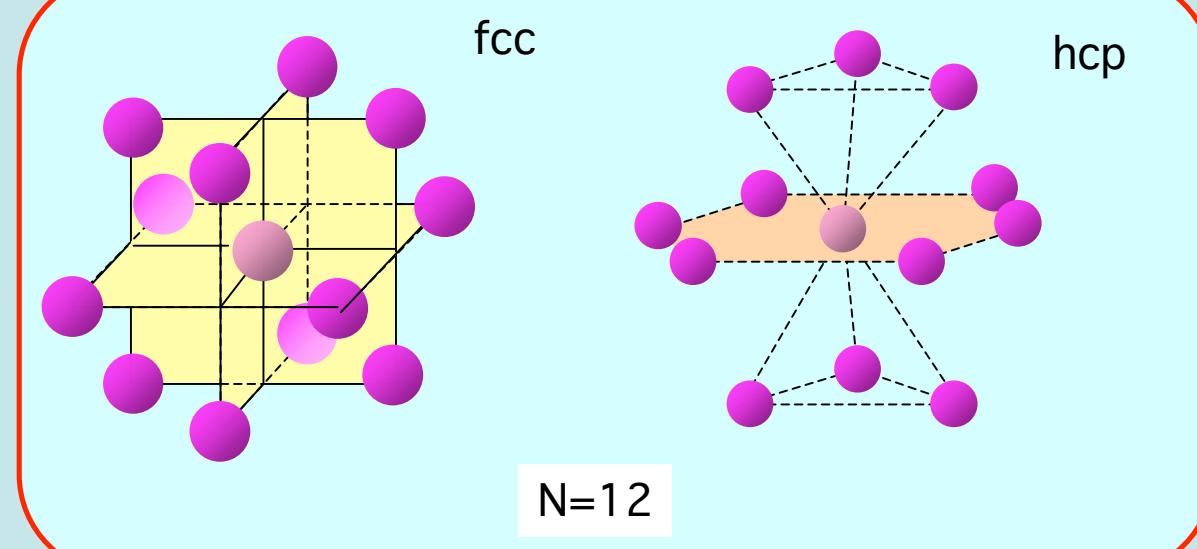
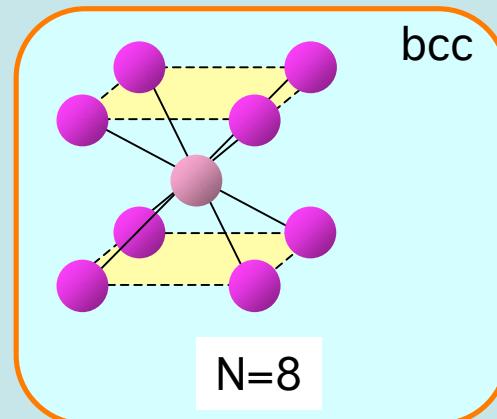
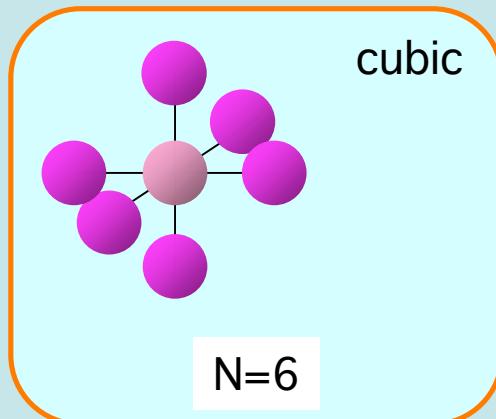
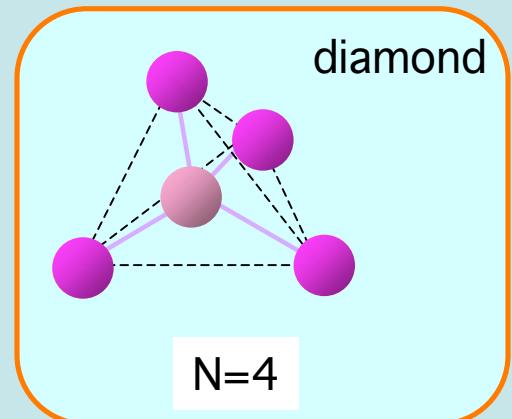
+ 2-atom basis



Coordination number = 12

Coordination number

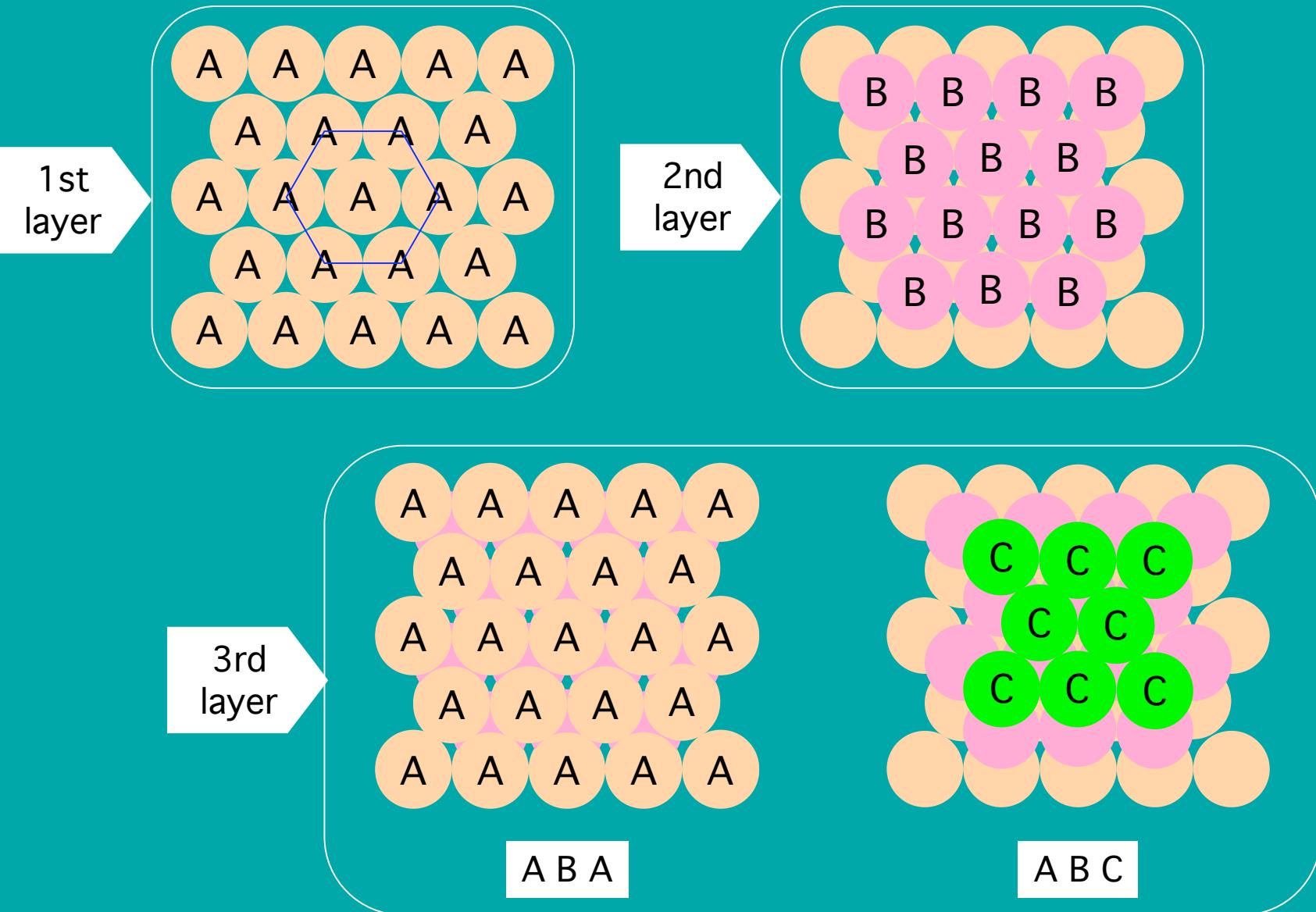
Paolo
Fornasini
Univ. Trento



Close packing

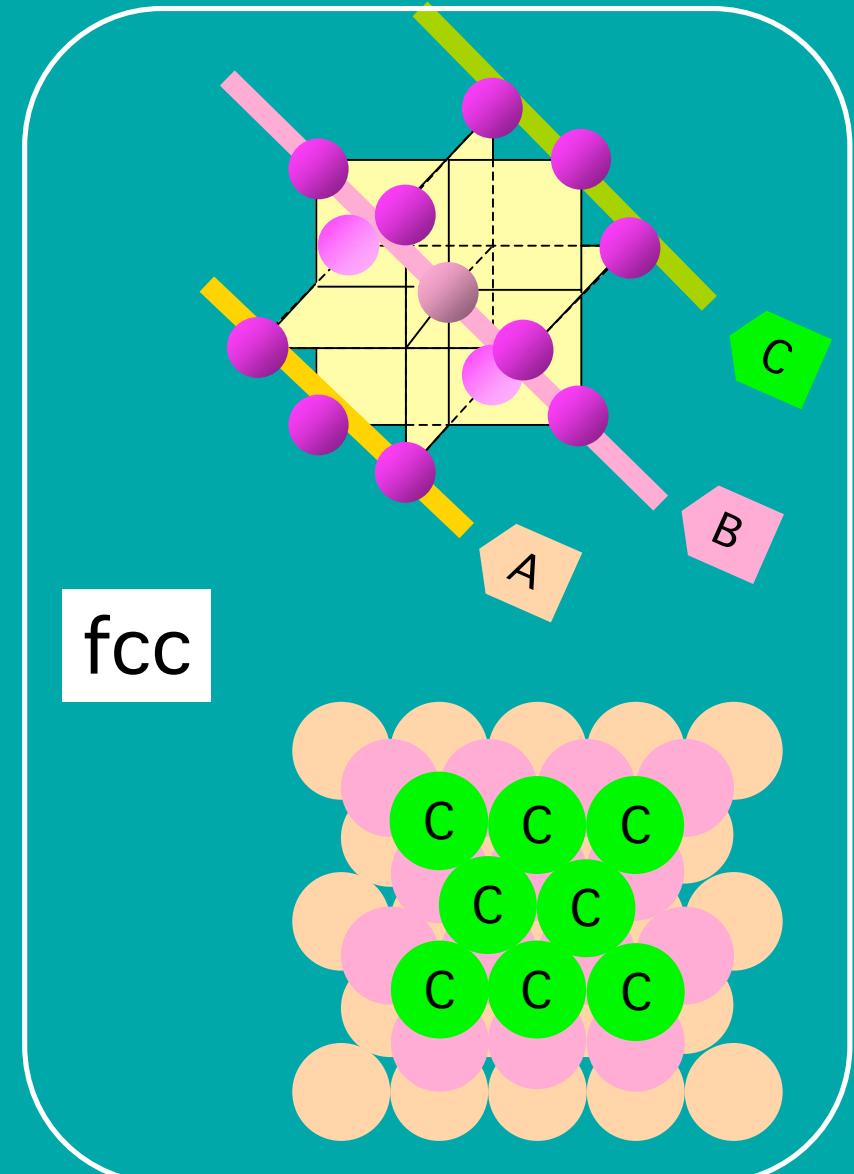
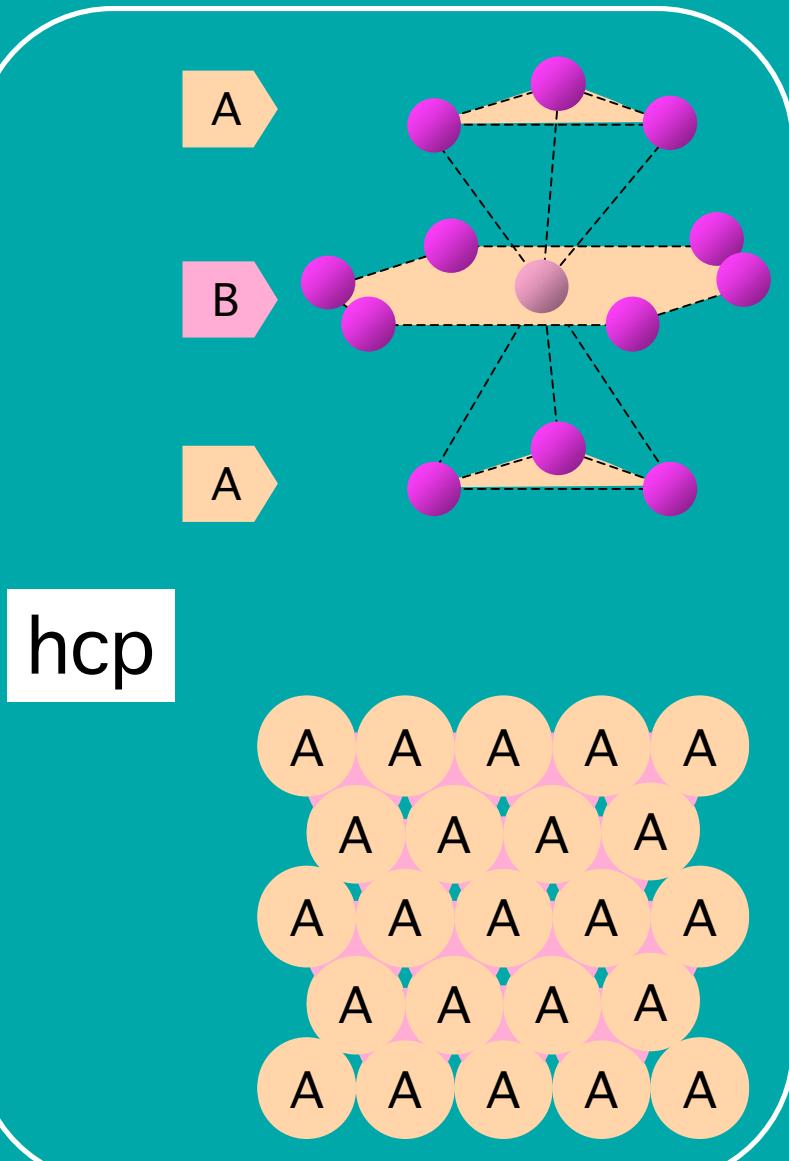
Close-packing of spheres

Paolo
Fornasini
Univ. Trento



hcp versus fcc

Paolo
Fornasini
Univ. Trento



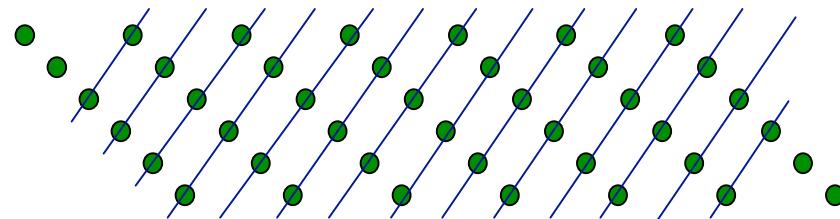
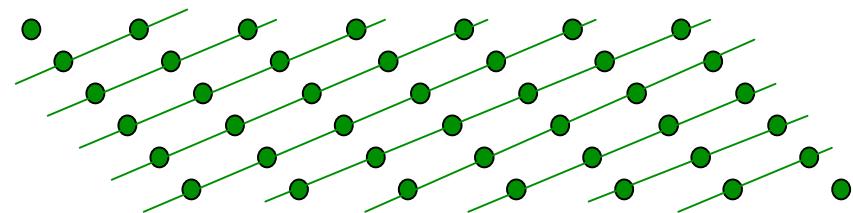
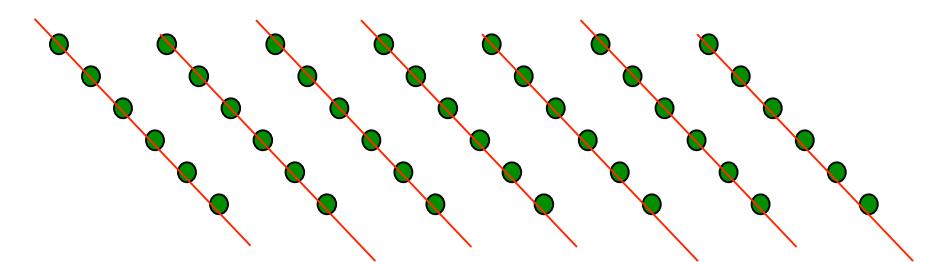
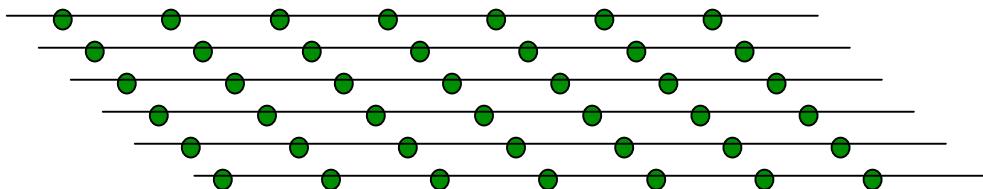


Crystal planes

Crystal planes

Paolo
Fornasini
Univ. Trento

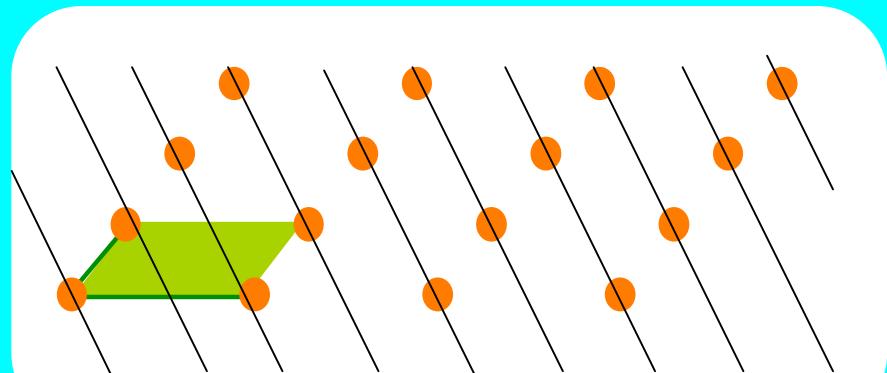
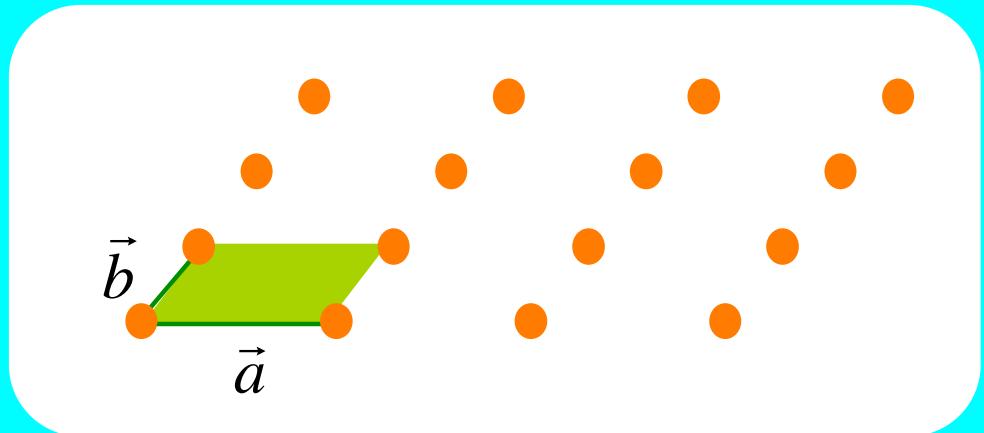
2-D



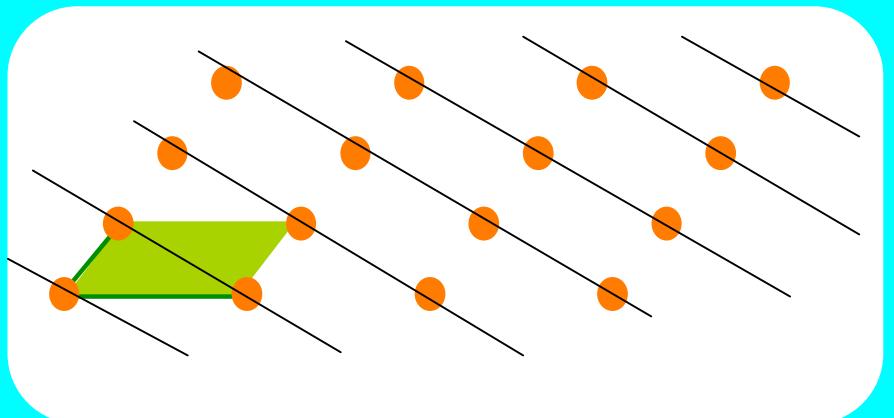
Miller indices, 2-D (a)

Paolo
Fornasini
Univ. Trento

2-D



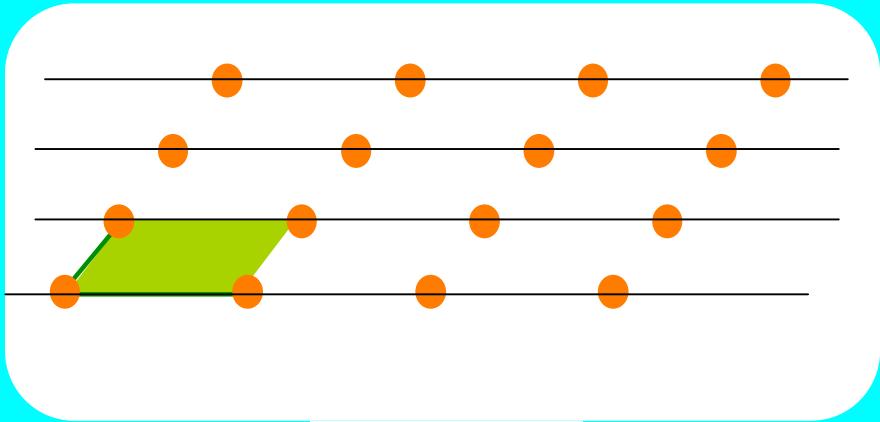
$(hk) = (21)$



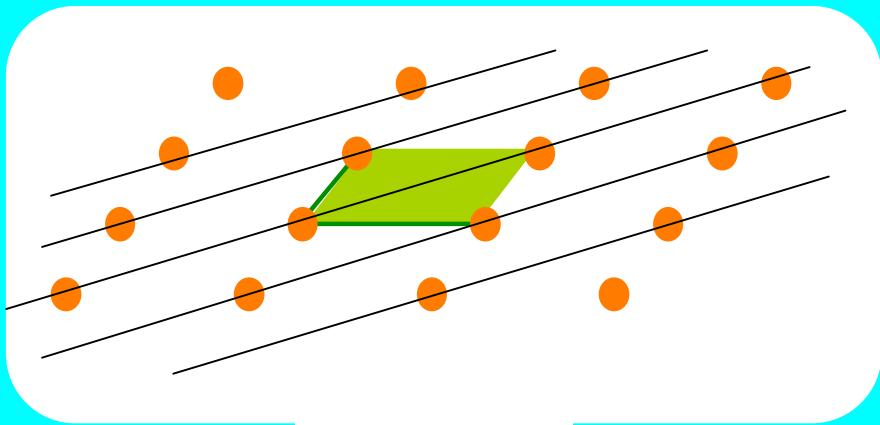
$(hk) = (11)$

Miller indices, 2-D (b)

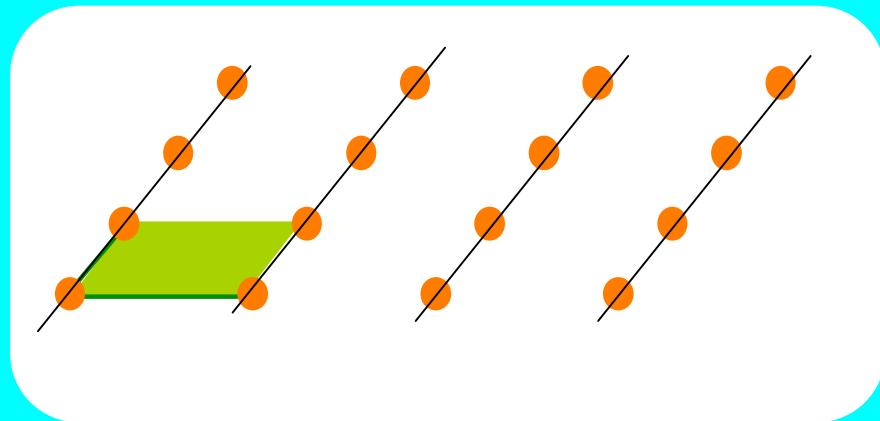
Paolo
Fornasini
Univ. Trento



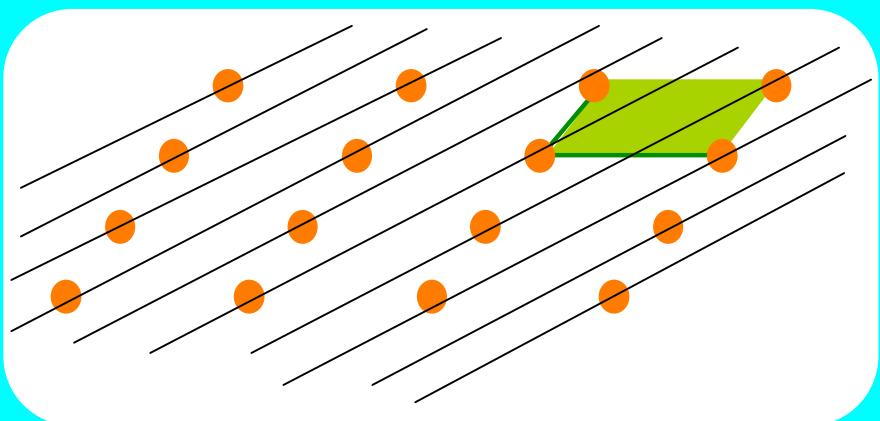
$$(hk) = (01)$$



$$(hk) = (1\bar{1})$$



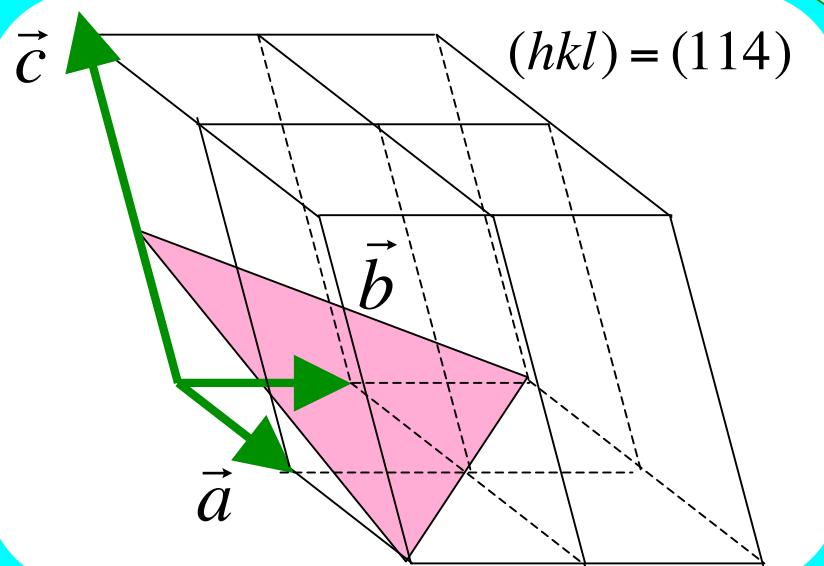
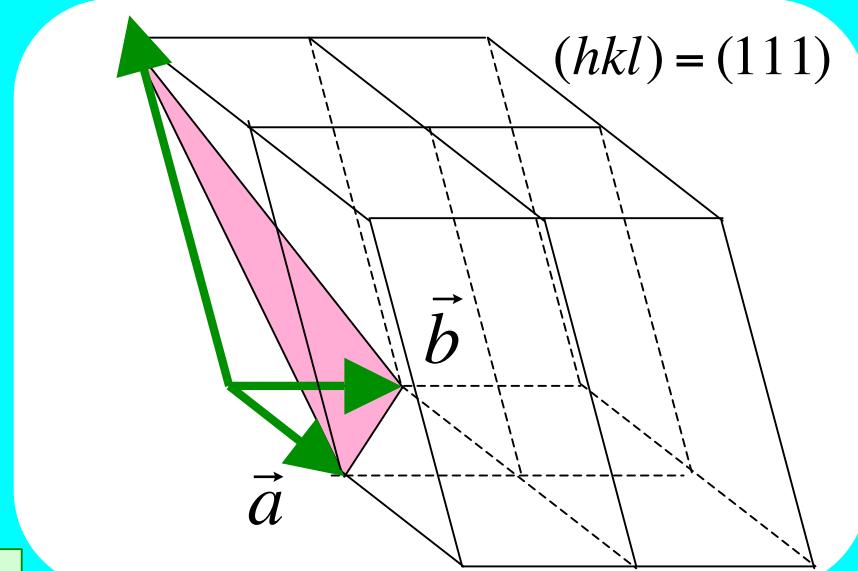
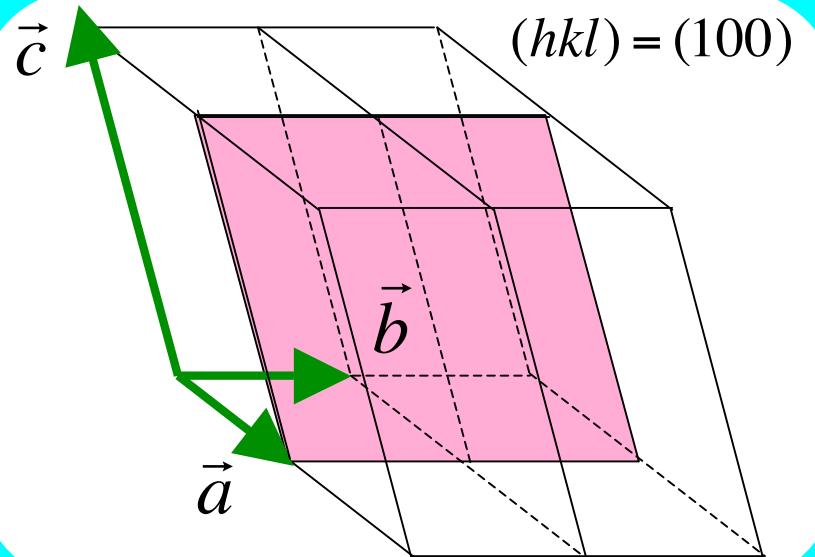
$$(hk) = (10)$$



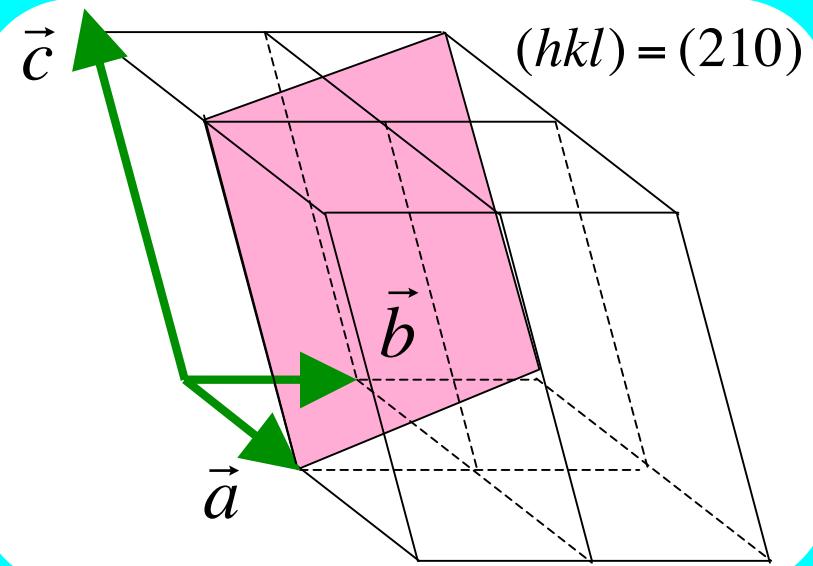
$$(hk) = (2\bar{1})$$

Miller indices, 3-D

Paolo
Fornasini
Univ. Trento



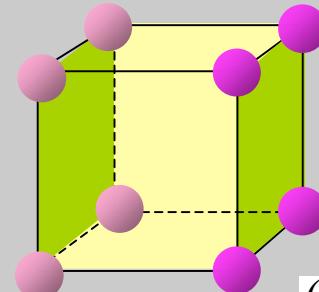
3-D



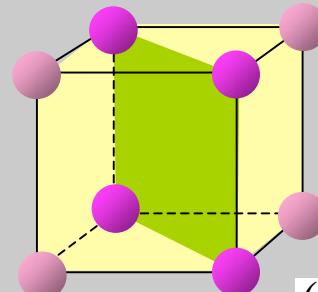
Miller indices, cubic lattices

Paolo
Fornasini
Univ. Trento

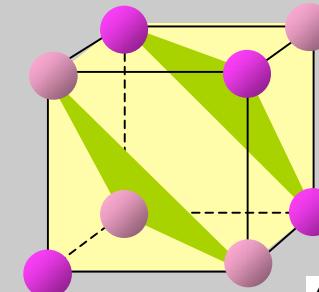
SC



(100)

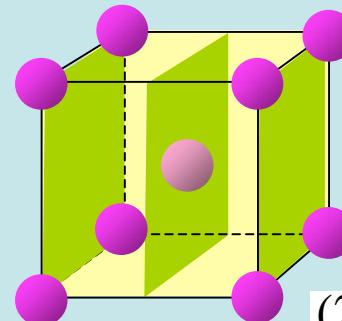


(110)

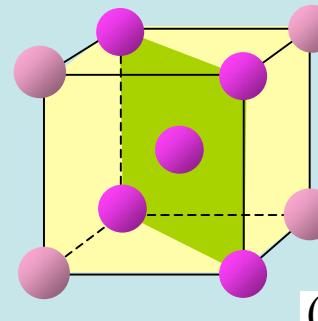


(111)

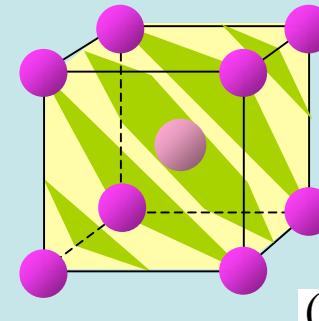
bcc



(200)

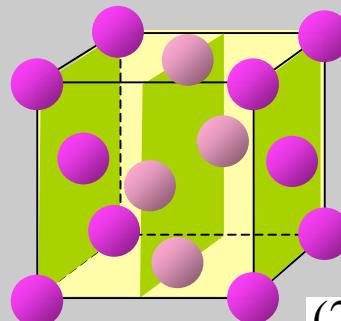


(110)

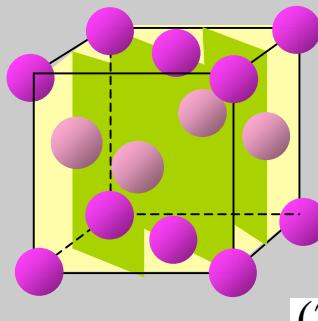


(222)

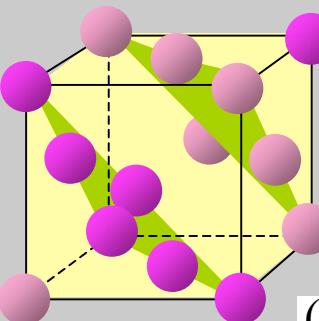
fcc



(200)



(220)



(111)

Interplanar distance

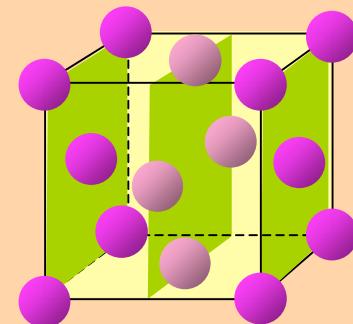
Paolo
Fornasini
Univ. Trento



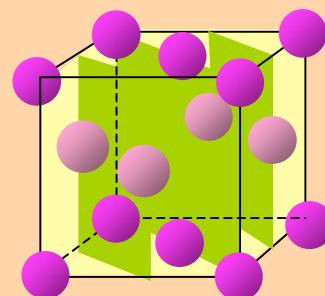
Cubic lattices

$$d_{hkl}^2 = \frac{a^2}{h^2 + k^2 + l^2}$$

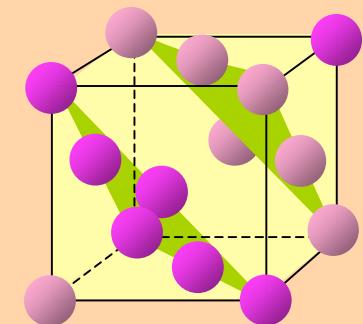
Copper, fcc, $a=3.61$ Å



$$d_{200} = \frac{a}{2} = 1.805 \text{ Å}$$



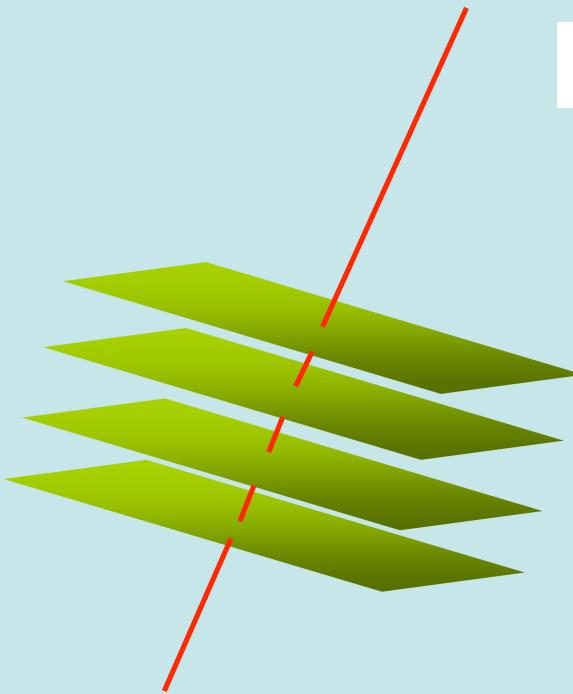
$$d_{220} = \frac{a}{2\sqrt{2}} = 1.276 \text{ Å}$$



$$d_{111} = \frac{a}{\sqrt{3}} = 2.084 \text{ Å}$$

Planes and directions

Family of planes

$$(hkl)$$


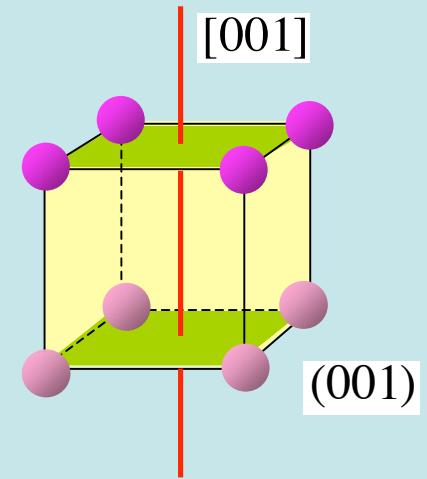
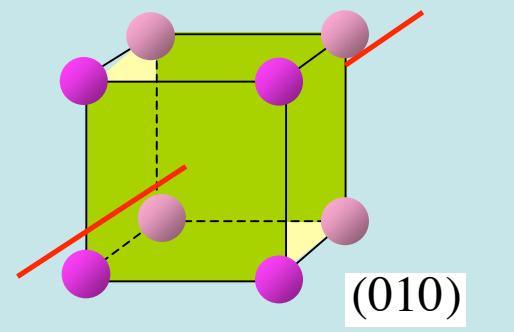
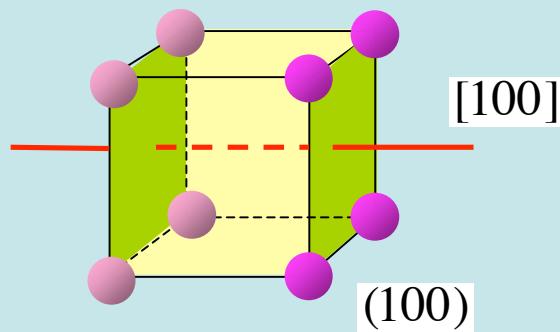
Perpendicular direction

$$[hkl]$$

Equivalent planes and directions

Equivalent directions

$\langle 100 \rangle$



Equivalent planes

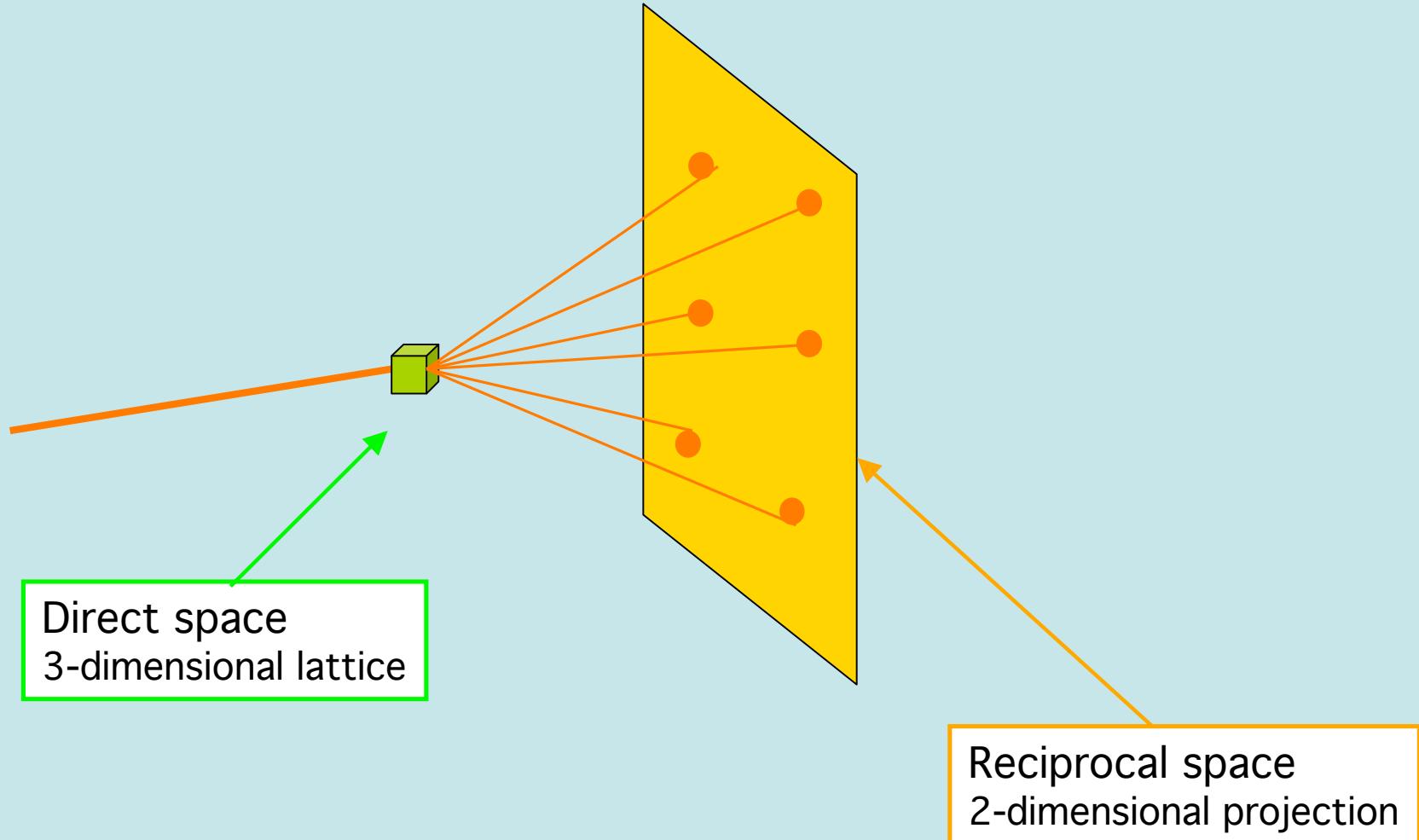
$\{100\}$



Reciprocal lattice

X-ray diffraction pattern

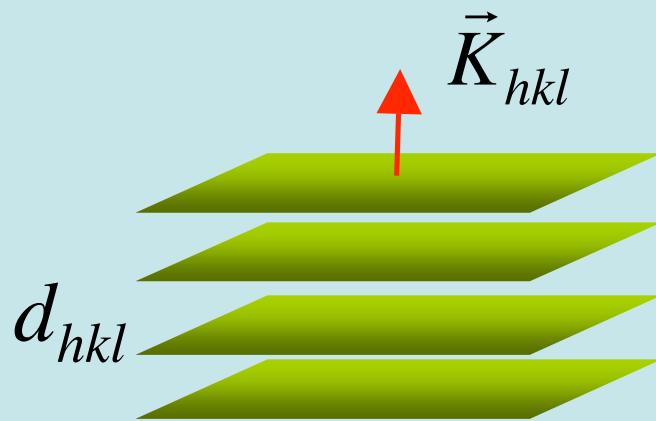
Paolo
Fornasini
Univ. Trento



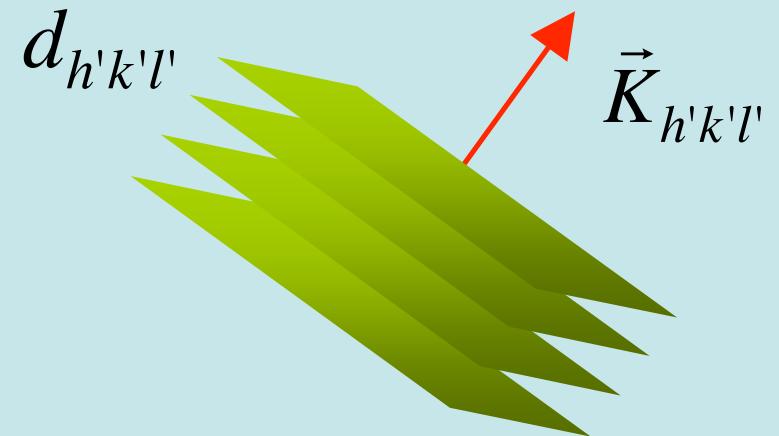
Basic idea

Paolo
Fornasini
Univ. Trento

A) Family of planes → wave-vector

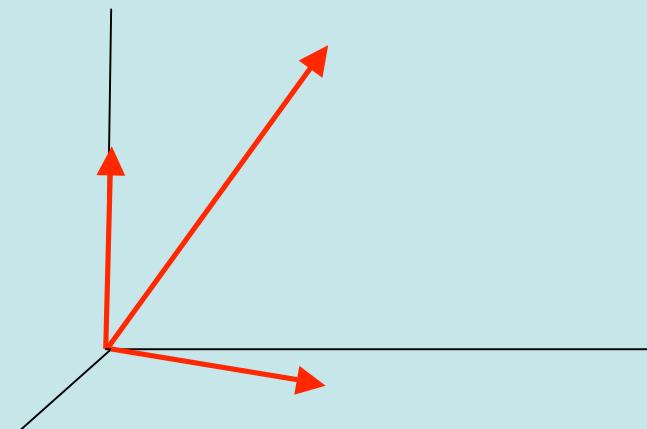


$$K_{hkl} = \frac{2\pi}{d_{hkl}}$$



B) Wave-vectors → set of points

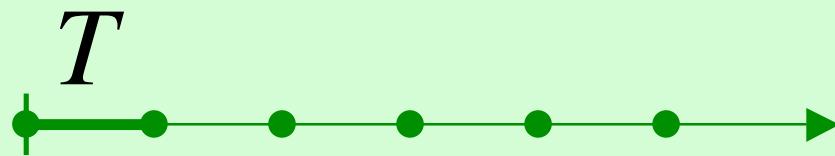
C) Set of points → lattice



Reciprocal quantities

Paolo
Fornasini
Univ. Trento

Periodic behaviour



time

$$\omega = 2\pi/T$$



frequency



position

$$k = 2\pi/\lambda$$

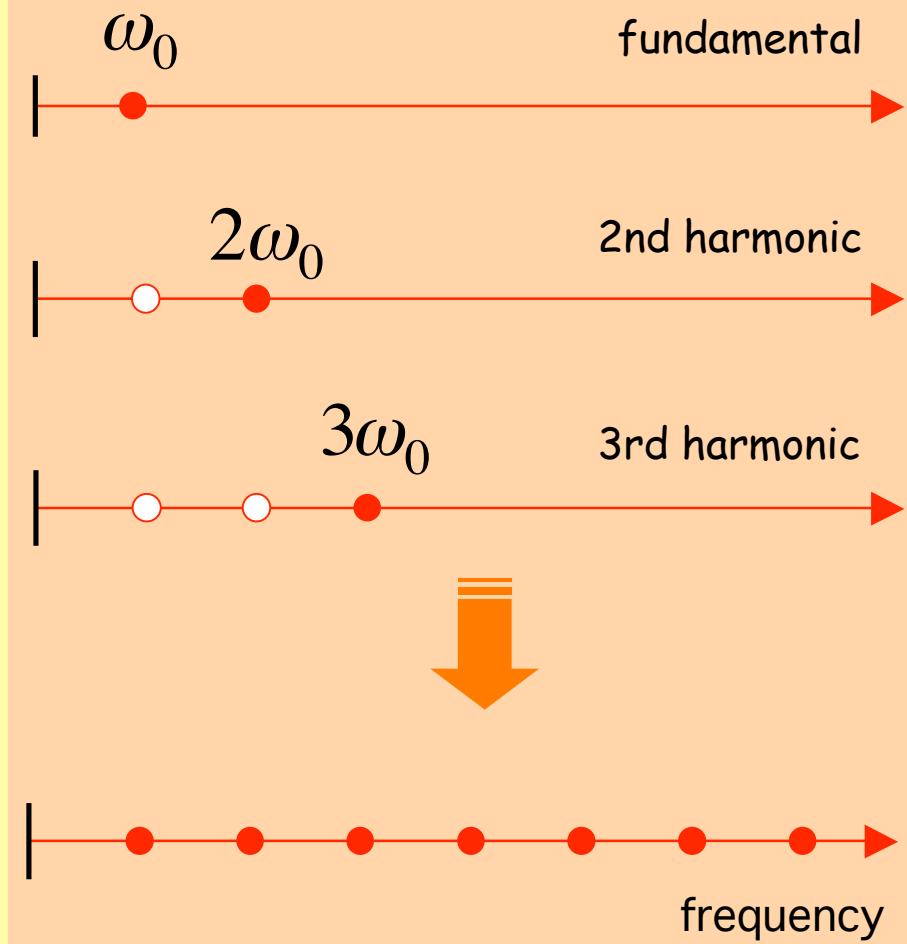
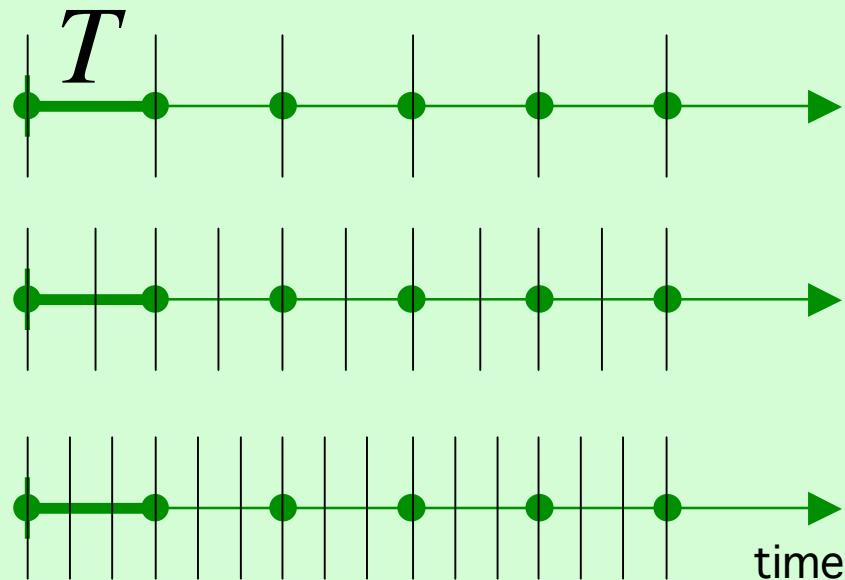


wave-vector

Harmonics

Paolo
Fornasini
Univ. Trento

Periodic behaviour

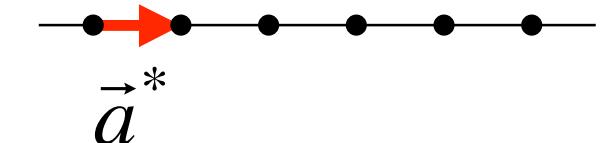
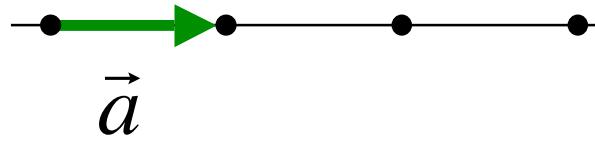
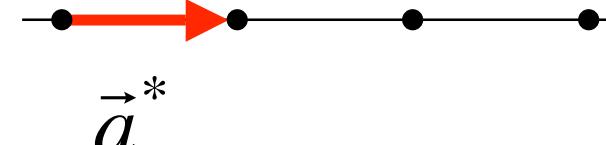
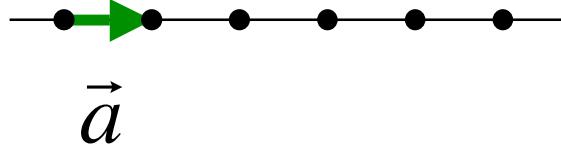


Periodic behaviour

Direct space

Reciprocal space

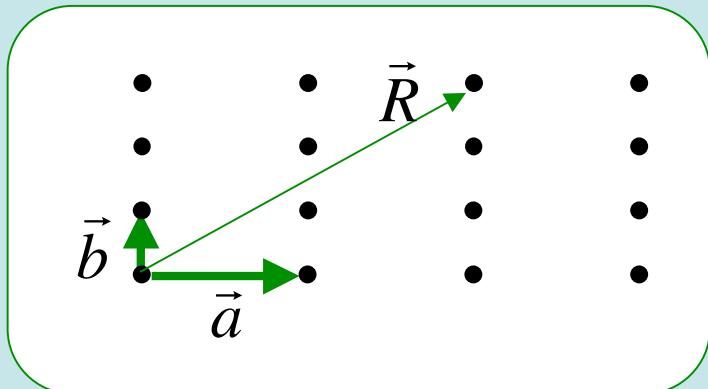
$$a^* = \frac{2\pi}{a}$$



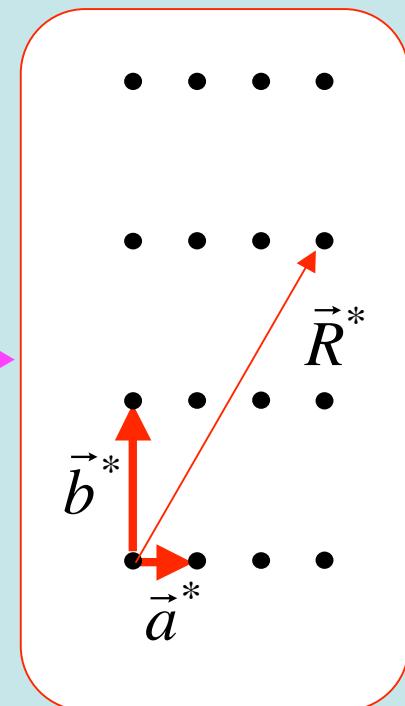
2-D, rectangular lattice (a)

Paolo
Fornasini
Univ. Trento

Direct space



Reciprocal space



$$a^* = \frac{2\pi}{a} = \frac{2\pi b}{ab}$$

$$b^* = \frac{2\pi}{b} = \frac{2\pi a}{ab}$$

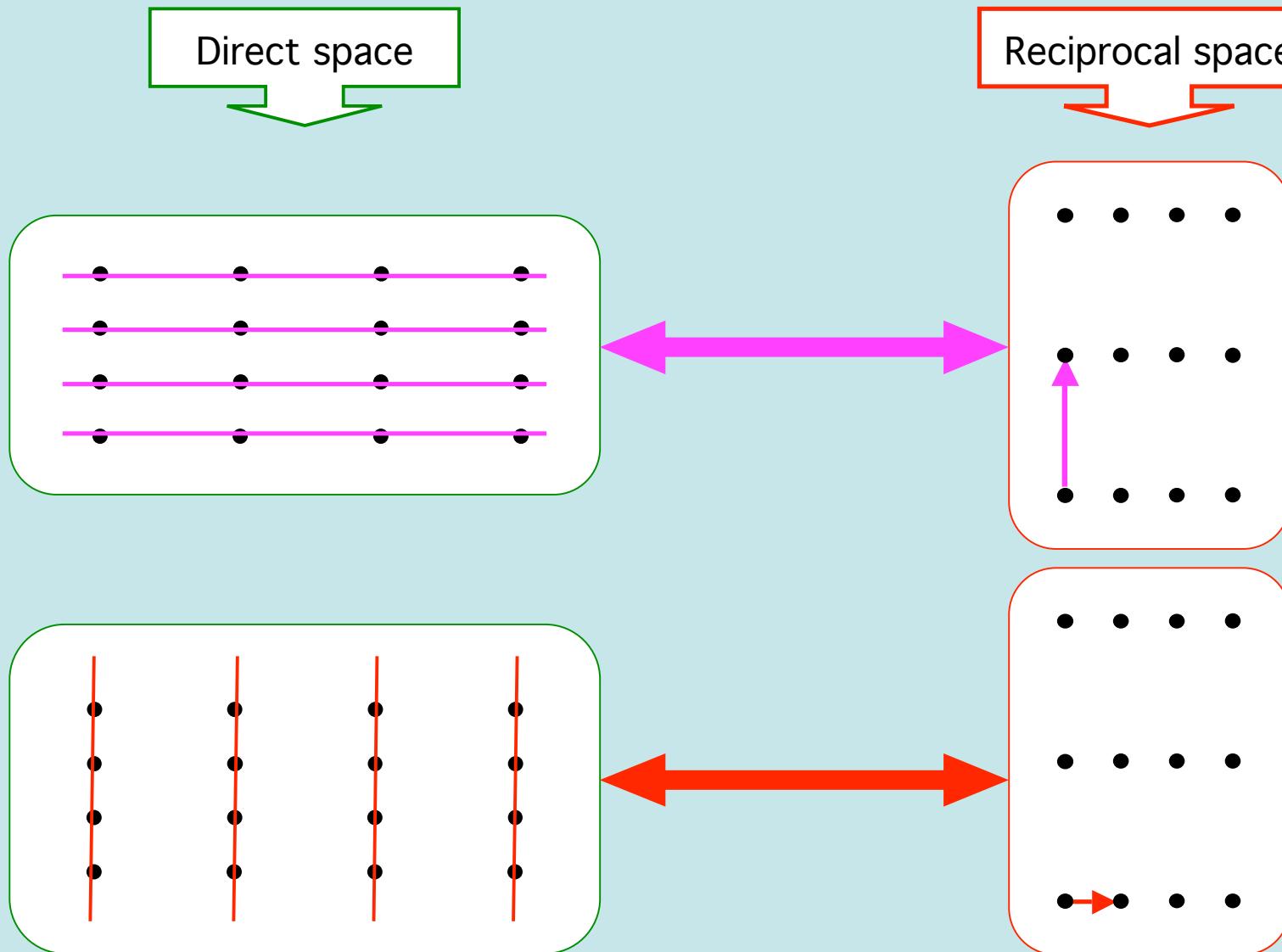
$$\vec{a}^* \perp \vec{b}$$
$$\vec{b}^* \perp \vec{a}$$

$$\vec{R} = n_1 \vec{a} + n_2 \vec{b}$$

$$\vec{R}^* = m_1 \vec{a}^* + m_2 \vec{b}^*$$

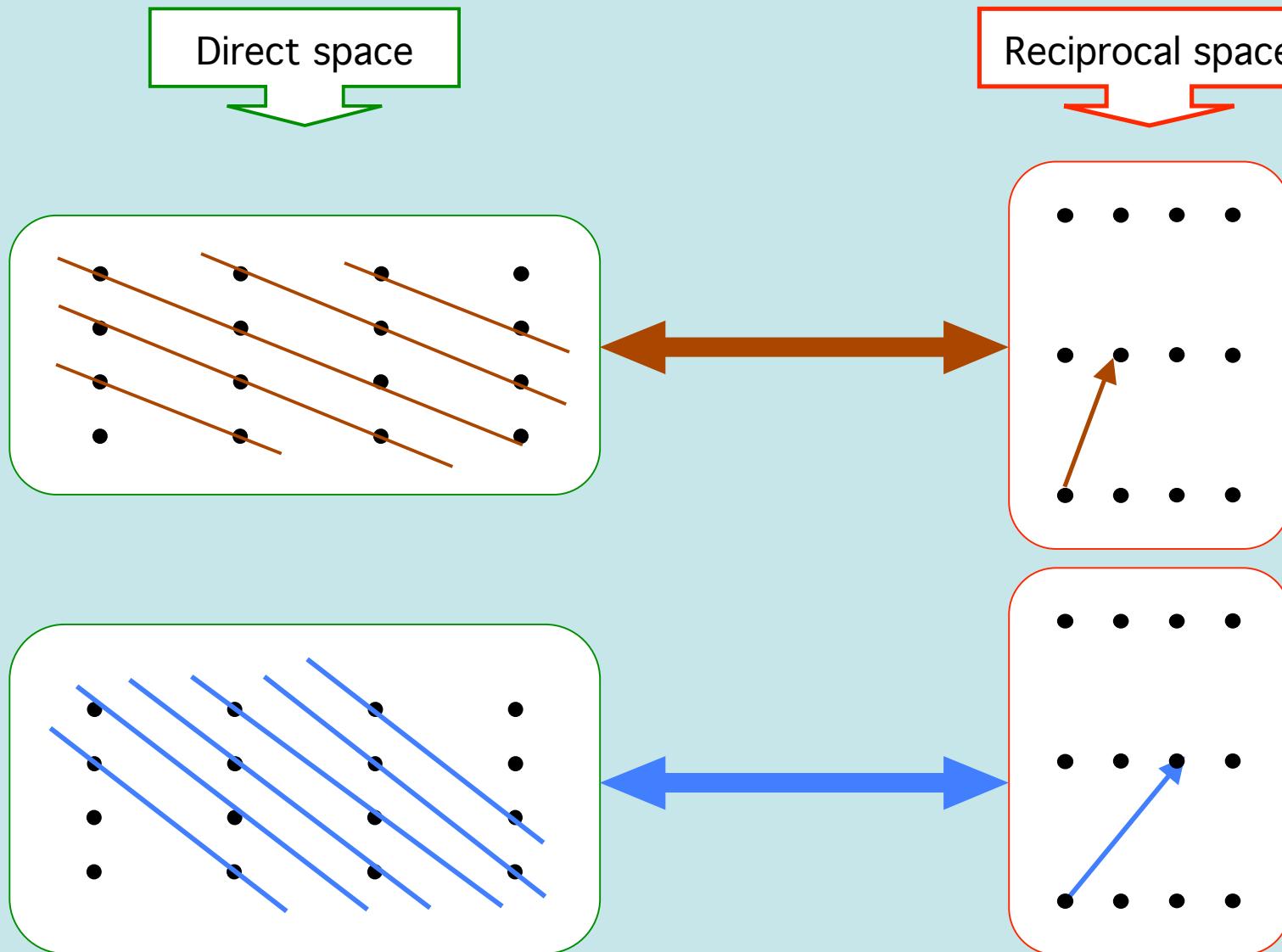
2-D, rectangular lattice (b)

Paolo
Fornasini
Univ. Trento



2-D, rectangular lattice (c)

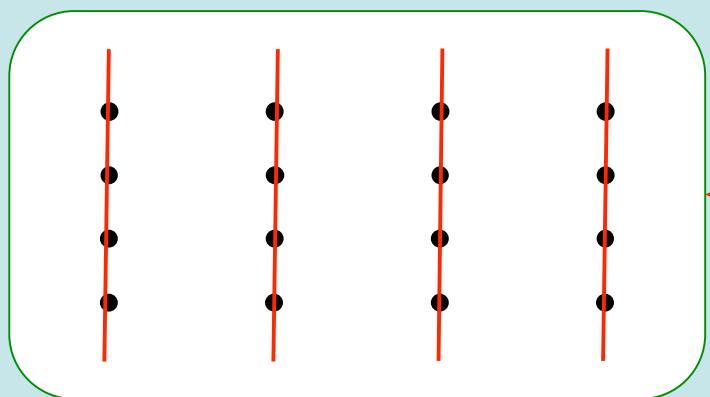
Paolo
Fornasini
Univ. Trento



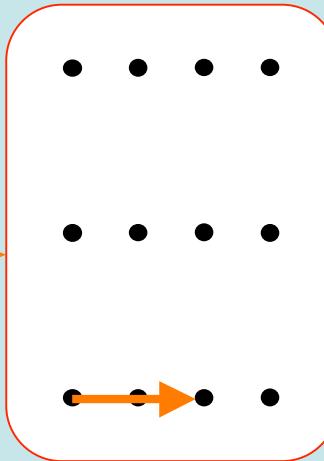
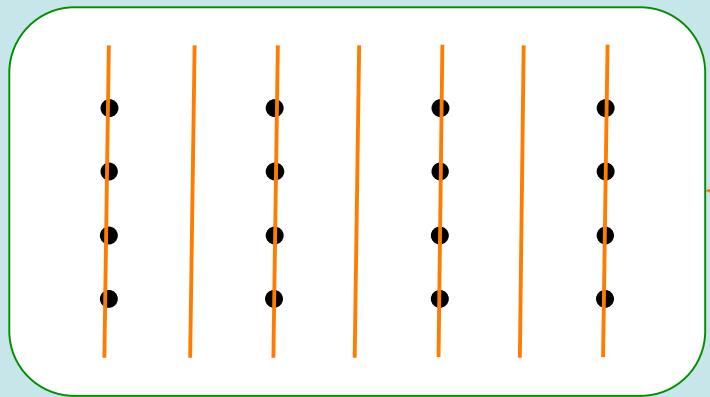
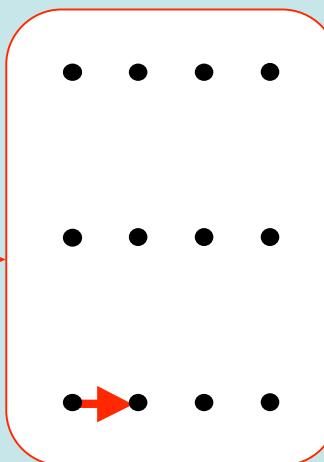
2-D, rectangular lattice (d)

Paolo
Fornasini
Univ. Trento

Direct space

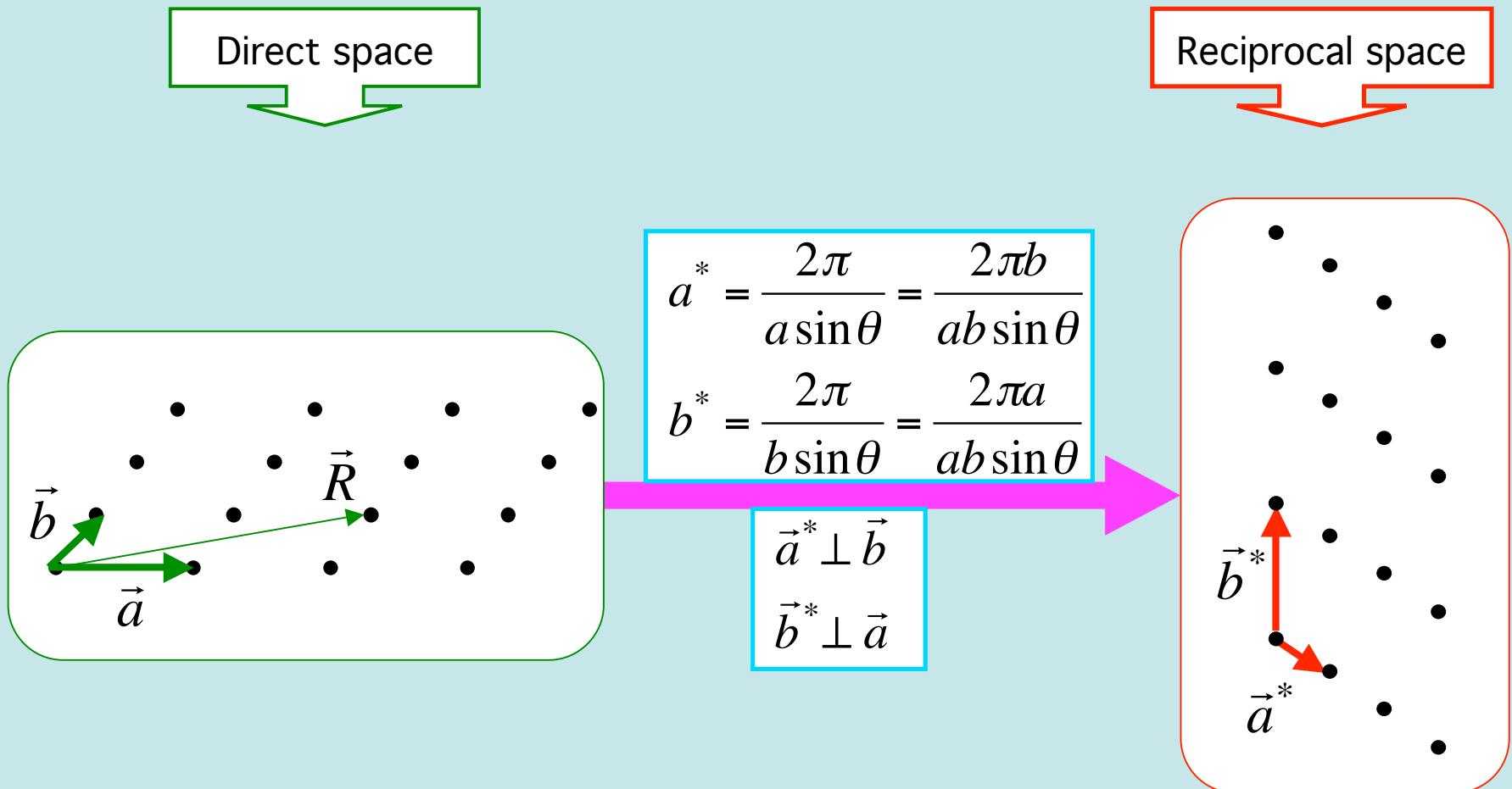


Reciprocal space



2-D, oblique lattice (a)

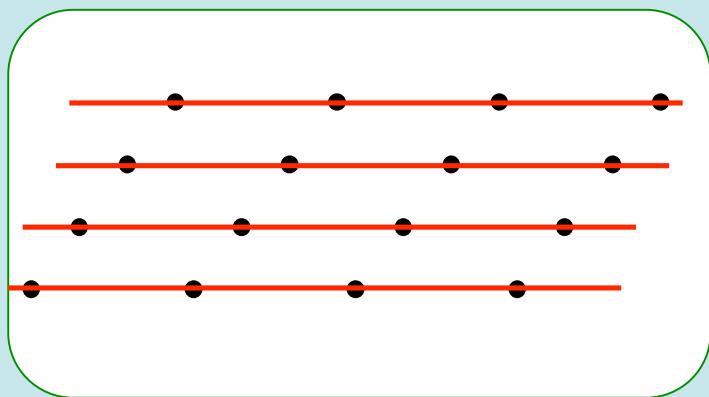
Paolo
Fornasini
Univ. Trento



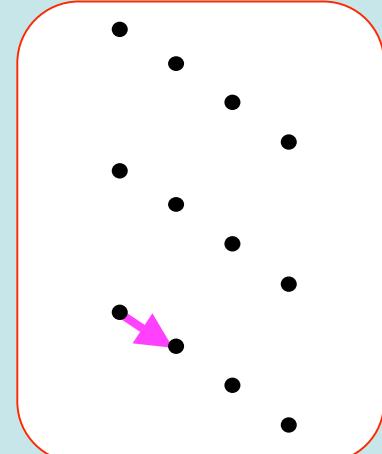
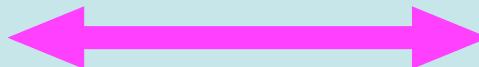
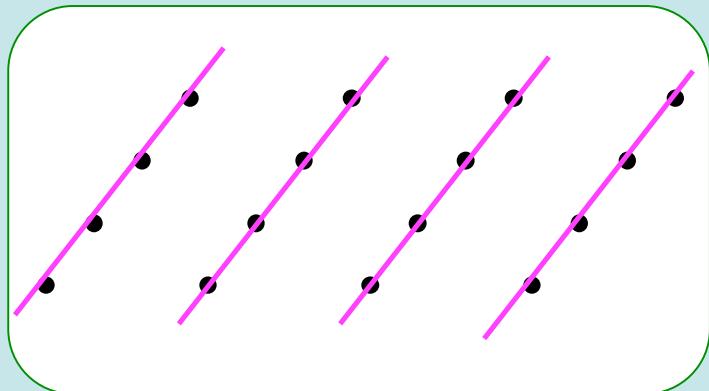
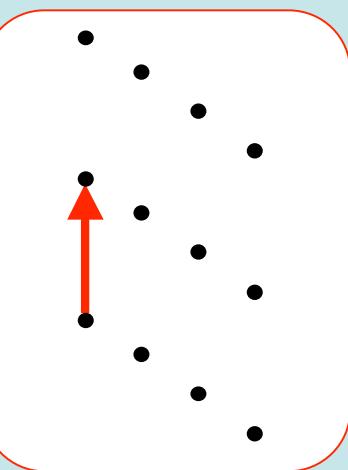
2-D, oblique lattice (b)

Paolo
Fornasini
Univ. Trento

Direct space

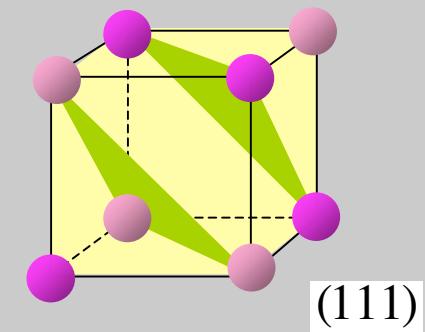
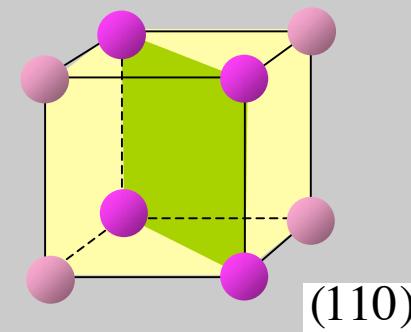
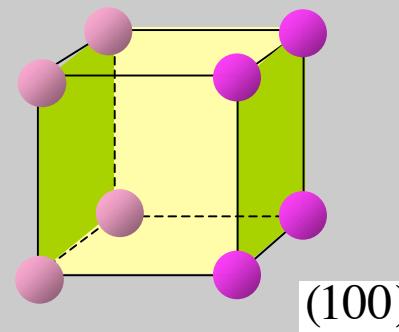
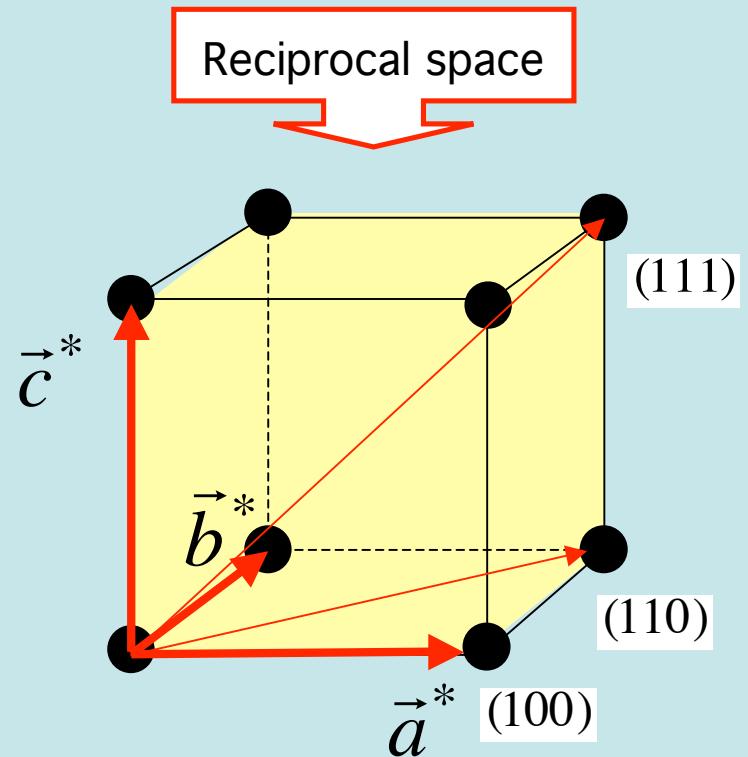
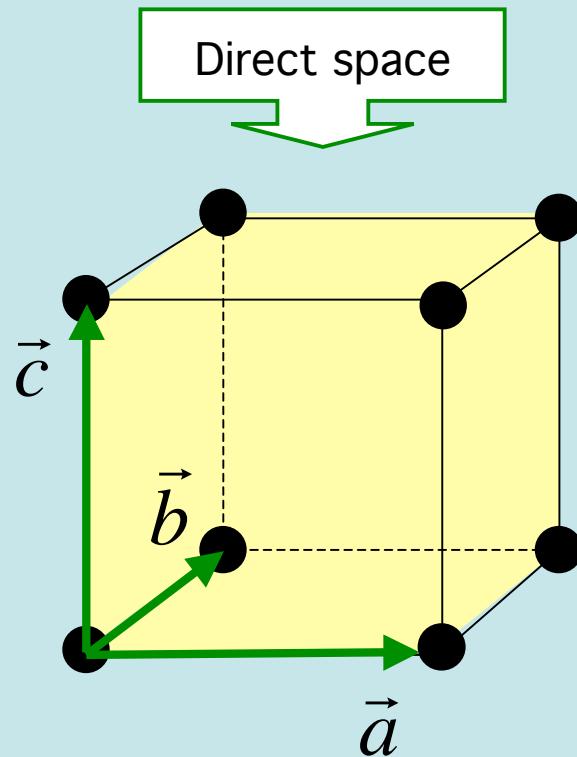


Reciprocal space



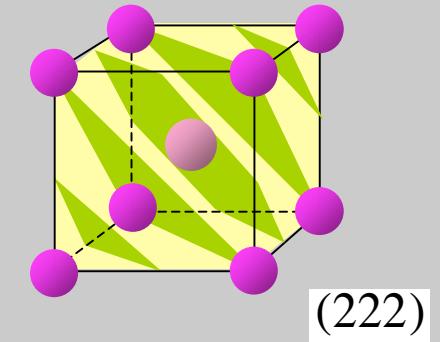
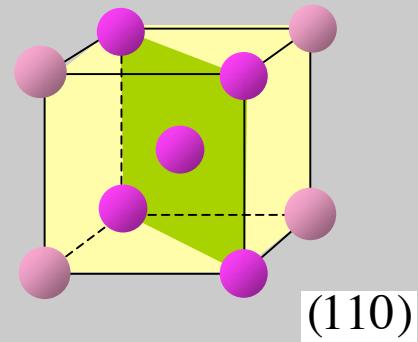
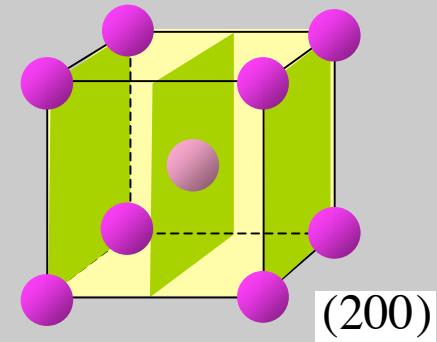
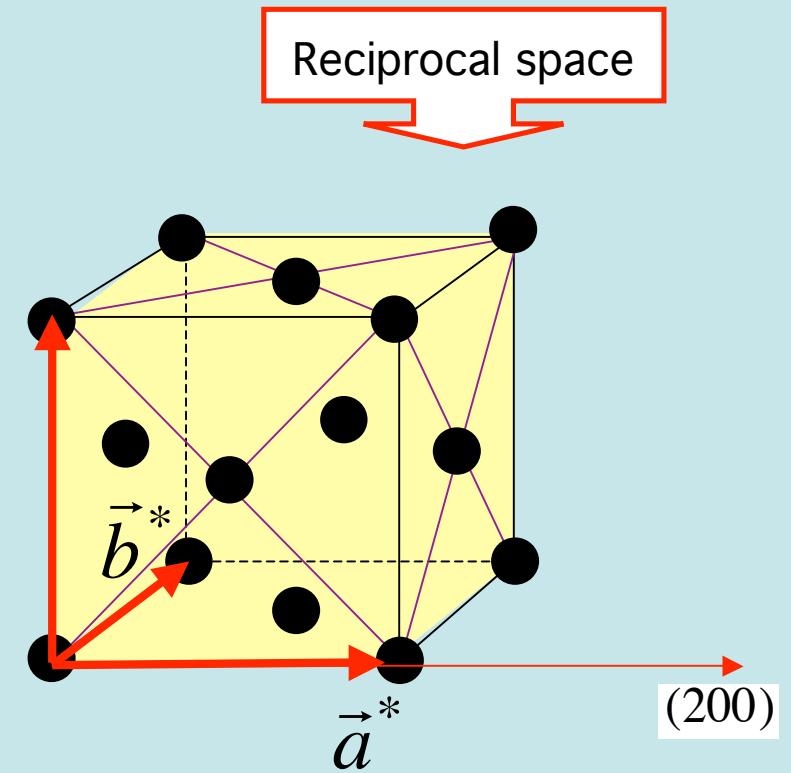
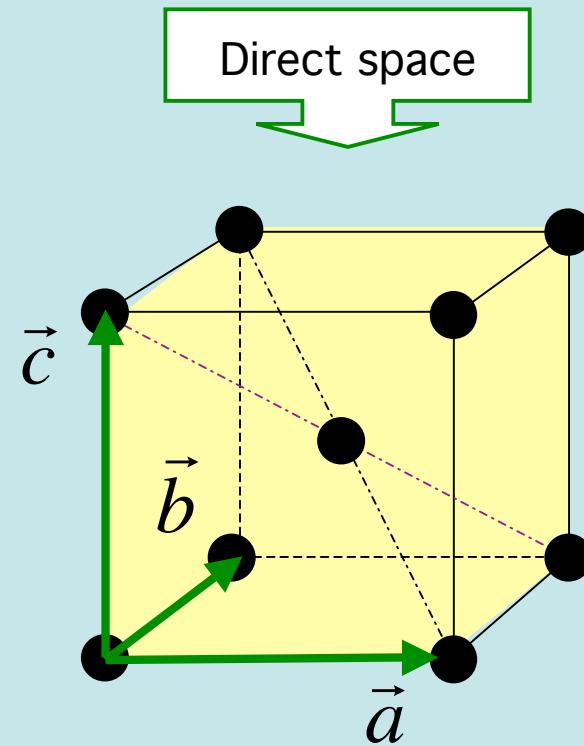
Cubic lattices (a)

Paolo
Fornasini
Univ. Trento



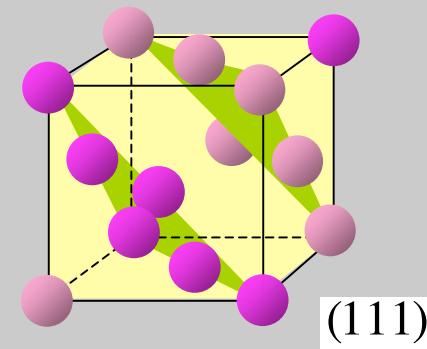
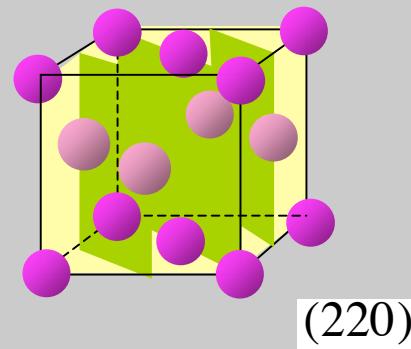
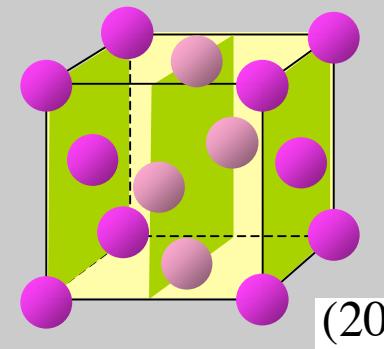
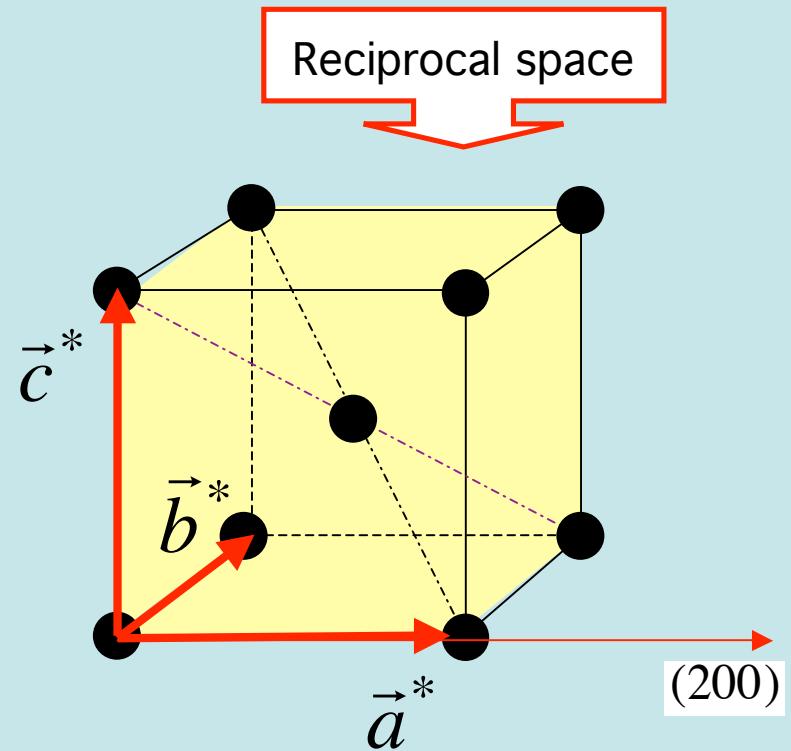
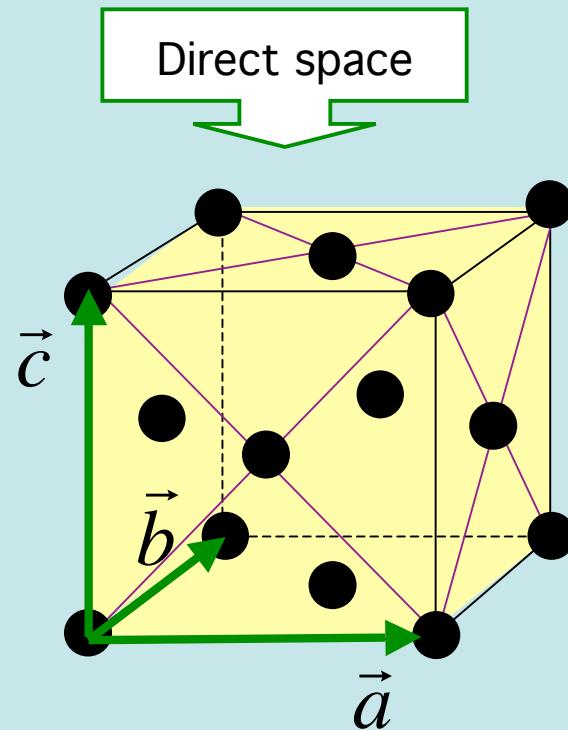
Cubic lattices (b)

Paolo
Fornasini
Univ. Trento



Cubic lattices (c)

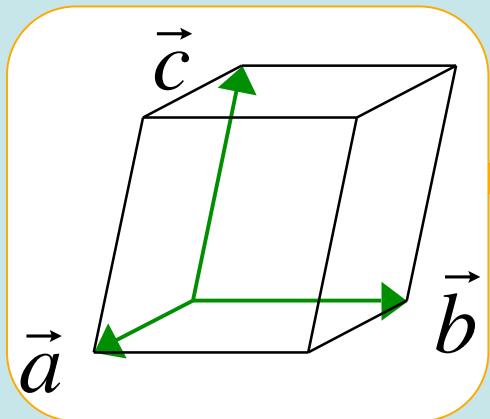
Paolo
Fornasini
Univ. Trento



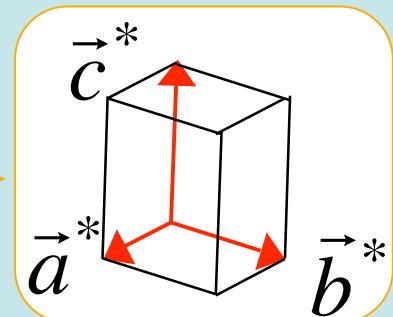
Primitive vectors: general rule

Paolo
Fornasini
Univ. Trento

Direct space



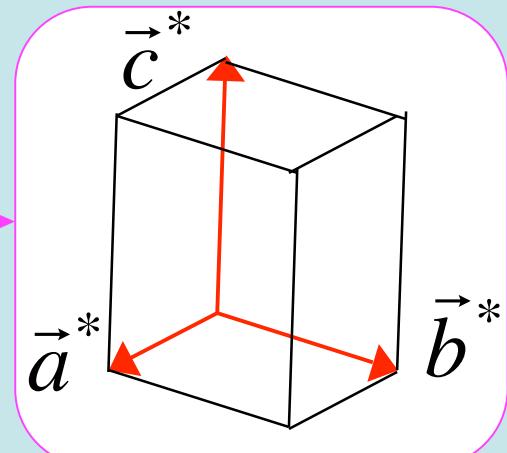
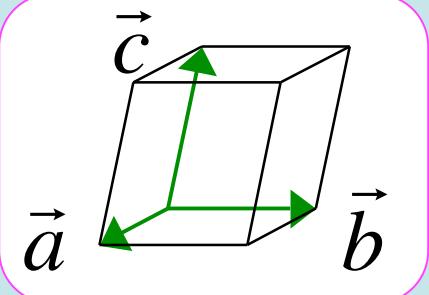
Reciprocal space



$$\vec{a}^* = 2\pi \frac{\vec{b} \times \vec{c}}{\vec{a} \cdot (\vec{b} \times \vec{c})}$$

$$\vec{b}^* = 2\pi \frac{\vec{c} \times \vec{a}}{\vec{a} \cdot (\vec{b} \times \vec{c})}$$

$$\vec{c}^* = 2\pi \frac{\vec{a} \times \vec{b}}{\vec{a} \cdot (\vec{b} \times \vec{c})}$$



$$\vec{a} \cdot (\vec{b} \times \vec{c})$$

Reciprocal lattice and lattice planes

Paolo
Fornasini
Univ. Trento

For any family of lattice planes separated by a distance d there are reciprocal lattice vectors perpendicular to the planes, the shortest of which have a length $2\pi/d$.

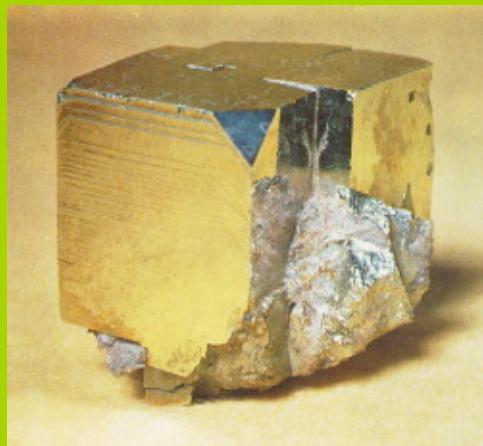
For any reciprocal lattice vector R^* , there is a family of lattice planes normal to R^* and separated by a distance d , where $2\pi/d$ is the length of the shortest reciprocal lattice vector parallel to R^* .



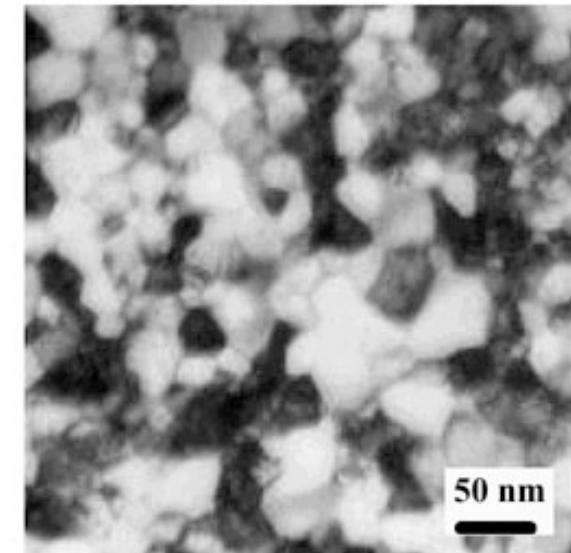
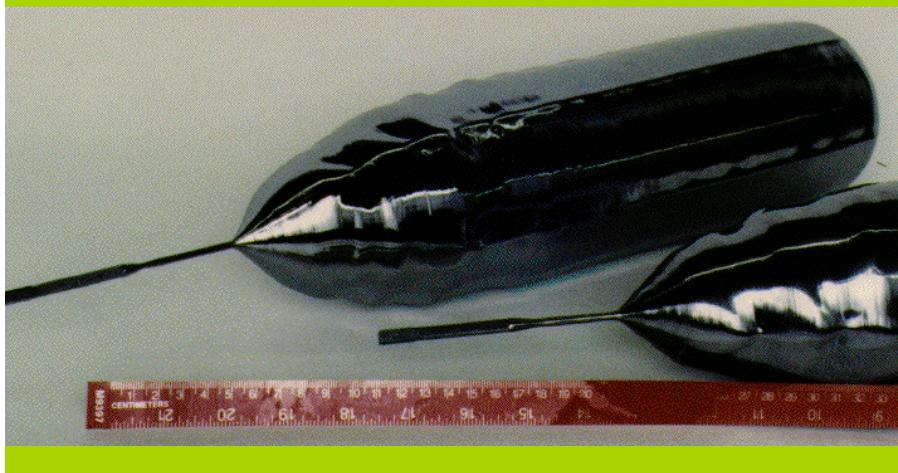
Microscopic structure of materials

Macro and micro-crystals

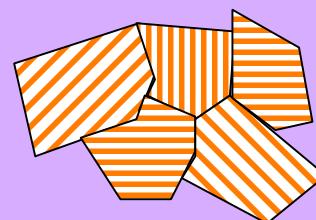
Paolo
Fornasini
Univ. Trento



Monocrystalline silicon, \varnothing 13 cm



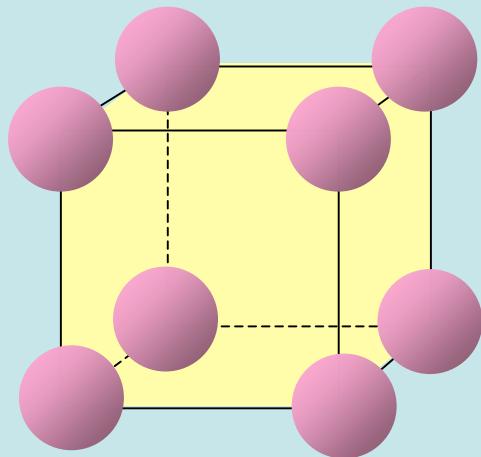
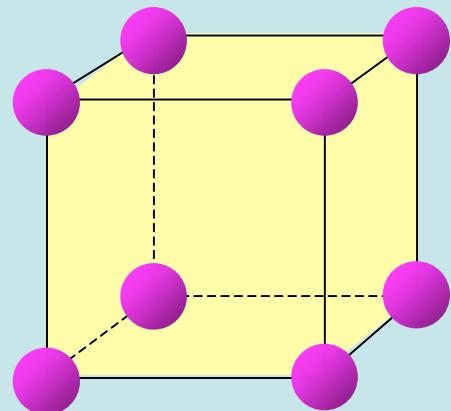
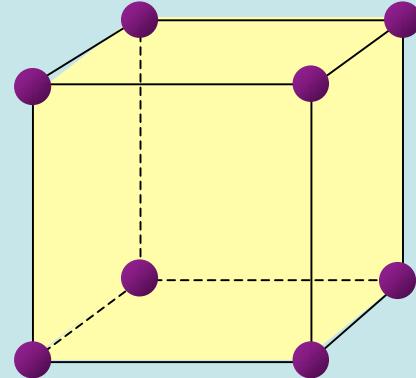
Cr, electron microscopy



Grain structure

Effects of temperature

Paolo
Fornasini
Univ. Trento



Temperature

Thermal motion

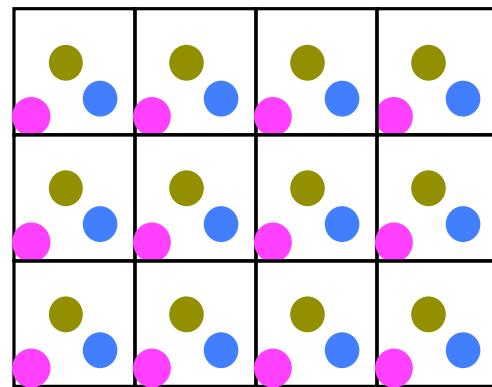


Spread of atomic positions

Crystalline and non-crystalline materials

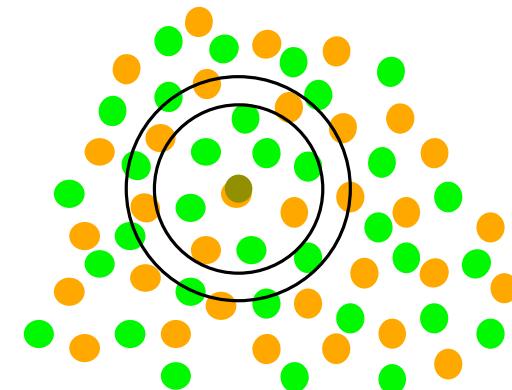
Paolo
Fornasini
Univ. Trento

Crystalline solids



Long-range order

Non-crystalline systems



No long-range order

Cooling rate

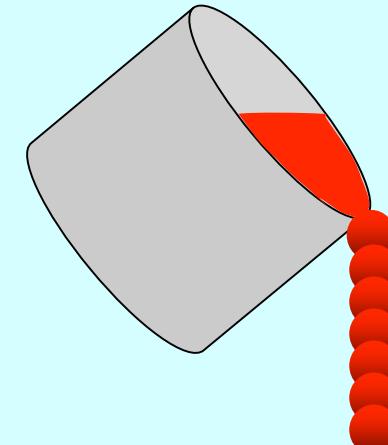
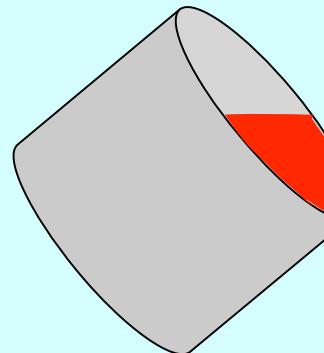
Paolo
Fornasini
Univ. Trento

High T:
liquid

Low T:
solid

Slow cooling

Fast cooling

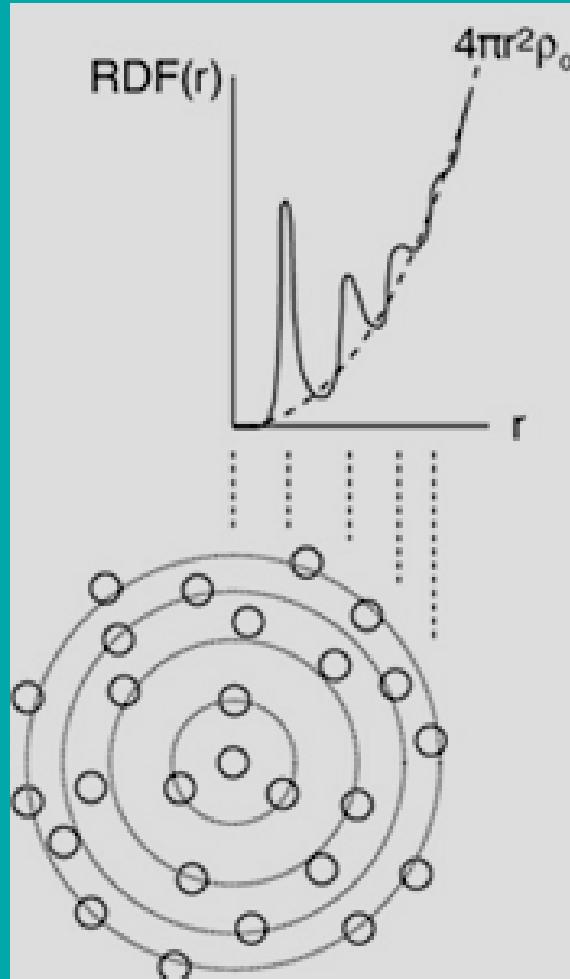


Thermodynamic
equilibrium

No thermodynamic
equilibrium

Radial Distribution Function

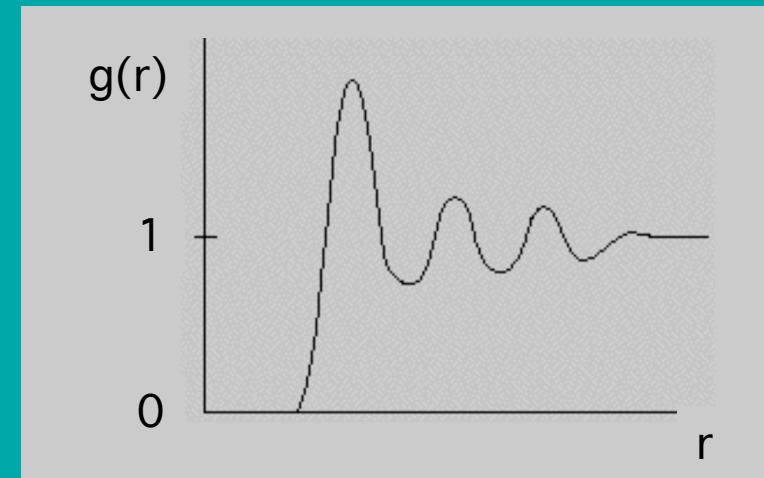
Paolo
Fornasini
Univ. Trento



$$\text{RDF} = 4\pi r^2 \rho(r) = 4\pi r^2 \rho_0 g(r)$$

average density

PDF = Pair Distribution Function



Short-range order



Summary

- Plane waves and wavevector
- Crystal structure = Bravais lattice + basis
- Bravais lattices: primitive vectors, unit cells (primitive and conventional), classifications
- Crystal structures (sc, bcc, fcc, hcp ...)
- Crystal planes and Miller indices
- Reciprocal lattice
- Crystalline and non-crystalline materials



Dolomite mountains in winter